#### A Novel Low-Temperature Diffusion Aluminide Coating for Ultrasupercritical Coal-Fired Boiler Applications

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## Why Iron Aluminide Coatings?



- Improvement of coal-fired power plant efficiency requires increase in steam temperature & pressure
- Advanced 9%Cr ferritic/martensitic alloys may be creep resistant up to 650°C but they will suffer extensive steam-side oxidation

 Aluminide coatings have been shown to drastically reduce the oxidation rate in exhaust/steam environment

## **Effect of Aluminide Coatings**



- Unlike in dry air, many Fe-base alloys are rapidly oxidized in steam/humid air when not coated
- An environment of air + 10 vol.% H<sub>2</sub>O can be used as a low cost method for determining coating performance

### Aluminide Coatings Fabricated at 900-1050°C via Chemical Vapor Deposition or Pack Cementation



- Thick CVD or Pack Coatings: 250-300 μm (6h at 1050°C)
- Thin CVD Coatings: 50-100 μm (6h at 900°C)

#### High-temperature aluminide coatings showed good long-term oxidation protection in air + 10 vol.% H<sub>2</sub>O



- At 700°C, thick & thin aluminide coatings have passed 18,000h
   & 10,000h, respectively (Pint et al., submitted to Surf. Coat. Technol., 2007)
- The critical AI in the coating to form  $AI_2O_3$  was ~3.5 at.%

#### **Disadvantages of High Aluminizing Temperatures**



- Nearly all aluminizing processes were carried out at 900-1150°C for 4-16h
- Thermochemical treatment of ferritic/martensitic steels at these temperatures can severely degrade their mechanical properties (creep resistance) (Rohr et al., Mater. Corros. 2005)

#### **Research Focus**

- Synthesize pack aluminide coatings at low-temperatures (≤ 700°C) on ferritic/martensitic alloys with reduced brittleness
- Low aluminizing temperature will
  - ensure no  $M \rightarrow F$  phase transformation
  - prevent increase of the grain size of the substrate alloy
  - reduce manufacturing cost if combining coating process with heat-treatment cycle
- Limitations of Low Aluminizing Temperature
  - Slow coating growth due to low Al vapor pressure & slow diffusion
  - Tendency to form brittle Al-rich phases

### **Three Key Research Components**

Task 1: Fabrication of Low-Temp. Aluminide Coatings1.1 Selection of Substrate Alloys1.2 Aluminizing Process Optimization1.2.1 Thermodynamic Calculations1.2.2 Aluminizing Process Optimization1.3 Coating Characterization

Task 2: Performance of Low-Temp. Aluminide coatings

- 2.1 Oxidation Resistance in Water-Vapor Environments
- 2.2 Coating Compositional & Microstructural Evolution during Thermal Exposure

Task 3: Effect of Aluminide Coatings on Mechanical Properties of Substrate Alloys

- 3.1 Creep Test
- 3.2 Cyclic Thermo-Mechanical Loading Test

# Pack cementation — Commercially Viable and Cost-Effective Method for Coating Fabrication



Reaction for AI Deposition: M (masteralloy) +  $AIX_x = MX_x$  (g) + AI

 Binary masteralloys can be used to lower the Al activity & reduce tendency of forming brittle Al-rich phases (Fe<sub>2</sub>Al<sub>5</sub>, FeAl<sub>3</sub>)

### **Thermodynamic Considerations**



Nciri & Vandenbulcke, in Proc. 4th European Conference on Chemical Vapor Deposition, 1983

- For the AI-Cr binary alloy, the activities of both AI and Cr are a function of the AI content
- Vapor pressures of AI halides were calculated for packs containing AI-Cr masteralloys with various AI at 700°C

#### The AI activity in the pack process can be tailored by adjusting the AI content in the masteralloy

6NH<sub>4</sub>CI-20(AI-Cr)-74AI<sub>2</sub>O<sub>3</sub> (in wt.%), 700°C





- AICI and AICI<sub>2</sub> are the species directly responsible for AI deposition (Xiang and Datta, J. Mater. Sci., 2005)
- When AI > 40 at.% in the AI-Cr masteralloys, AI deposition becomes dominant

## **Experimental Approaches**

•	Substrates: Commercial	Eleme
	Ferritic/Martensitic Alloy P91	Fe
	– Fe-9Cr-1Mo steel (wt.%)	Cr
•	Pack Cementation: 6-12 h at	Мо
	650°C and 700°C	Mn
•	Pack Powder Mixture	V
	1 2 wt % NH Clastivator	Si
	-1-2 WL 70 NH <sub>4</sub> CI activator	Ni
	– 10-20% masteralloy	Cu
	<ul> <li>Balance Al<sub>2</sub>O<sub>3</sub> filler</li> </ul>	Nb
•	Oxidation Testing:	С
	– 650°C	Ν
	– 100-h cycles	0
		•

Elements	wt.%	at.%
Fe	88.46	87.75
Cr	9.26	9.87
Мо	0.96	0.55
Mn	0.47	0.47
V	0.23	0.25
Si	0.19	0.37
Ni	0.16	0.15
Cu	0.07	0.06
Nb	0.05	0.03
С	516 ppm	0.238
Ν	480 ppm	0.190
0	26 ppm	0.009
S	8 ppm	0.001

# Pure Al was used as the masteralloy to obtain the baseline coatings at 650°C



Pack Composition	Average Coating Thickness (μm)		
(wt.%)	Surface	Around Corner	
1NH <sub>4</sub> CI-10AI-89AI <sub>2</sub> O <sub>3</sub>	7-47	60	
2NH <sub>4</sub> CI-20AI-78AI <sub>2</sub> O <sub>3</sub>	32-79	190	

#### Brittle Fe<sub>2</sub>Al<sub>5</sub> coating was formed when pure Al was used as masteralloy

#### 2NH<sub>4</sub>CI-20AI-78AI<sub>2</sub>O<sub>3</sub> (wt.%) 650°C / 6h





#### **Modifications in the Pack Aluminizing Process**



#### Fe<sub>2</sub>Al<sub>5</sub>/FeAl two-layer coating was formed using Cr-25wt.%Al masteralloy at 700°C

#### 2NH<sub>4</sub>CI-15(Cr-25AI)-83Al<sub>2</sub>O<sub>3</sub> 12h at 700°C



 A layer of 4 μm Fe<sub>2</sub>Al<sub>5</sub> on top of 12 μm FeAl





# FeAl coating (~10μm) was obtained with reduced Al activity using Cr-15wt.%Al masteralloy

#### 2NH<sub>4</sub>CI-20(Cr-15AI)-78AI<sub>2</sub>O<sub>3</sub> 12h at 700°C







### Hardness of Fe<sub>2</sub>Al<sub>5</sub>, Fe<sub>2</sub>Al<sub>5</sub>/FeAl and FeAl Coatings



#### Initial Oxidation Results of Low-Temperature Pack Coatings in Air + 10 vol.% H<sub>2</sub>O at 650°C



# Low-temperature pack coatings mimic the as-coated morphology after 1,000h air + 10% H<sub>2</sub>O at 650°C



# Thin coatings (50-100 $\mu$ m) are preferred for concerns of CTE mismatch and creep resistance



The main degradation of mechanical strength of the coated alloy was due to the decrease in the load-bearing section because of the weak creep properties of Fe-Al compared to the P92

#### **Comparison of AI Reservoir in Aluminide Coatings**

Coating	Thickness (μm)	Surface / Interface Al (at.%)	Al Reservoir
Thin CVD Coating	50	18	450
LT Fe <sub>2</sub> Al <sub>5</sub> / FeAl coating	4 + 12	65 + 50	560
LT FeAl coating	10	40	200

- The thin CVD coating passed 10,000h at 700°C before failure
- Minimal interdiffusion is expected at 650°C (Zhang et al., *Mater. Corros.*, 2007, in press)
- Low-temp. pack coatings with Fe<sub>2</sub>Al<sub>5</sub>/FeAl should have a lifetime comparable to CVD thin coatings
- Brittle Fe<sub>2</sub>Al<sub>5</sub> could lead to cracking



### **Effect of the Amount of Masteralloy**



• The AI reservoir increased with the amount of materalloy

 When > 50 wt.% of masteralloy was used, the change of the Al profiles became insignificant

# Summary

- Non-uniform Fe<sub>2</sub>Al<sub>5</sub> coating was formed when pure AI was used as the masteralloy at 650°C
- The AI activity in the pack cementation process was reduced by using AI-containing binary masteralloys
  - With Cr-25Al masteralloy, a coating of Fe<sub>2</sub>Al<sub>5</sub>/FeAl was formed at 700°C
  - Cr-15Al masteralloy led to formation of a thin (10-12  $\mu$ m) FeAl coating
  - The AI reservoir increased with the amount (< 50%) of Cr-15AI materalloy at 700°C
- The aluminide coatings synthesized at 700°C showed good initial oxidation behavior at 650°C in air + 10% H<sub>2</sub>O

## **Future Work**

- **1.** Fabrication of Low-Temp. Aluminide Coatings
- 2. Performance of Low-Temp. Aluminide coatings
- 3. Effect of Aluminide Coatings on Mechanical Properties of Substrate Alloys: Creep Test in Steam Environment





- Increase the pack aluminizing temperature to 750°C
- Combine the coating process with standard heat treatment
- Apply surface mechanical treatment (shot peening) prior to coating processing (Xiang and Datta, Scipta Mater., 2006)

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