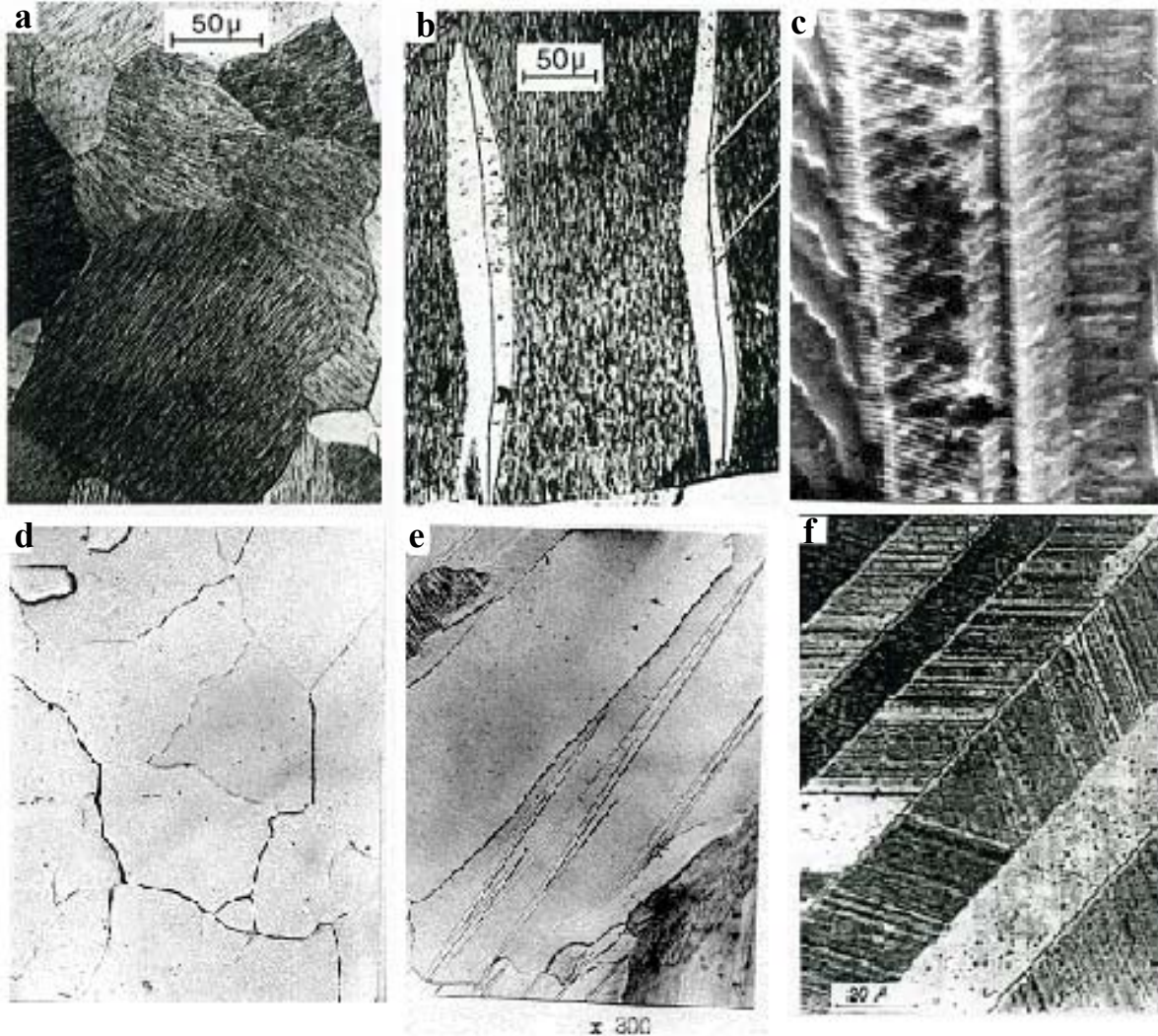


Influence of Hydrogen Content on Some Properties and Presence of Defects in Zirconium Hydride

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FSUE VNIIA named after N. L. Dukhov, Moscow

Zirconium hydride structures depending on hydrogen content 1



δ - phase
 $\text{ZrH}_{1.66}$

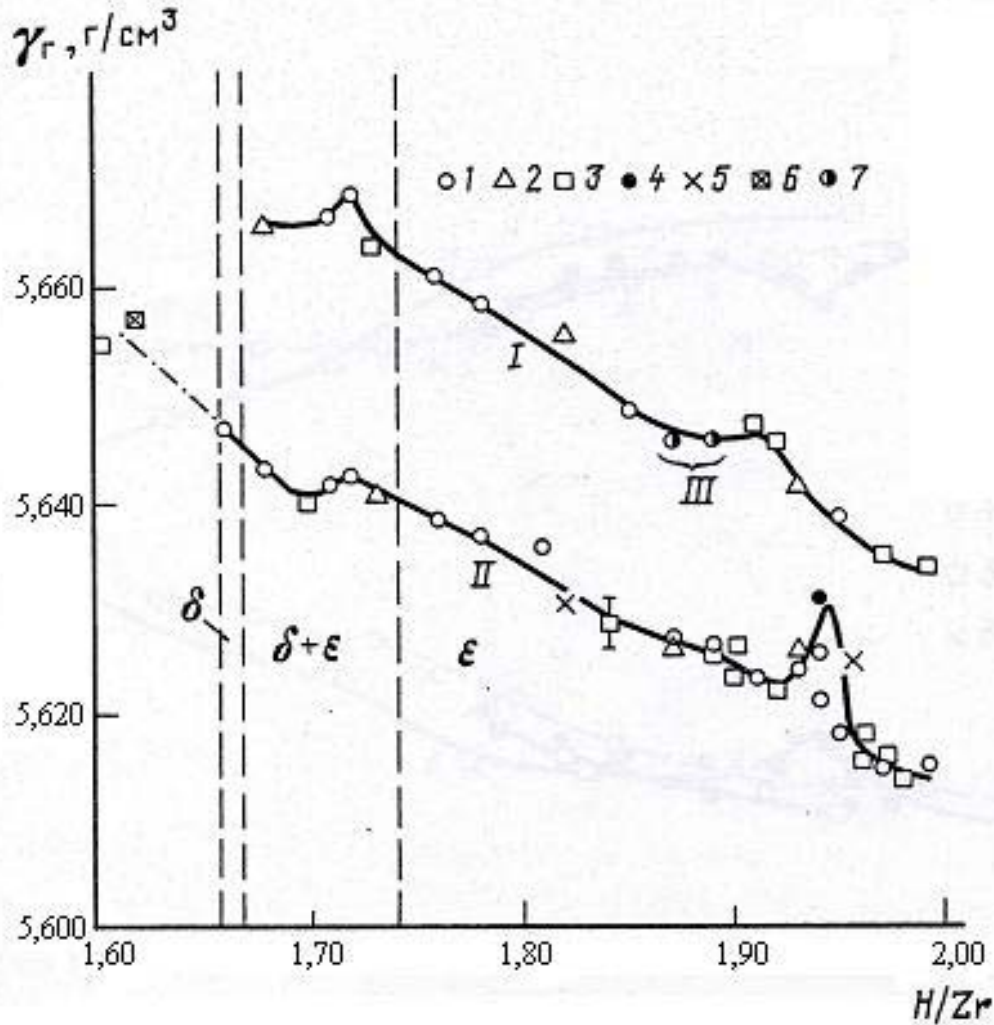
$\delta + \epsilon$ - phases
 $\text{ZrH}_{1.66-1.67}$

ϵ - phase
 $\text{ZrH}_{1.93}$

Concentration dependences of hydrostatic density of zirconium hydride samples

(I- doped zirconium 1.1% Nb, 0.07% O (mass.); II – undoped zirconium;

III - 1.1% Nb, 0.01% O **2**

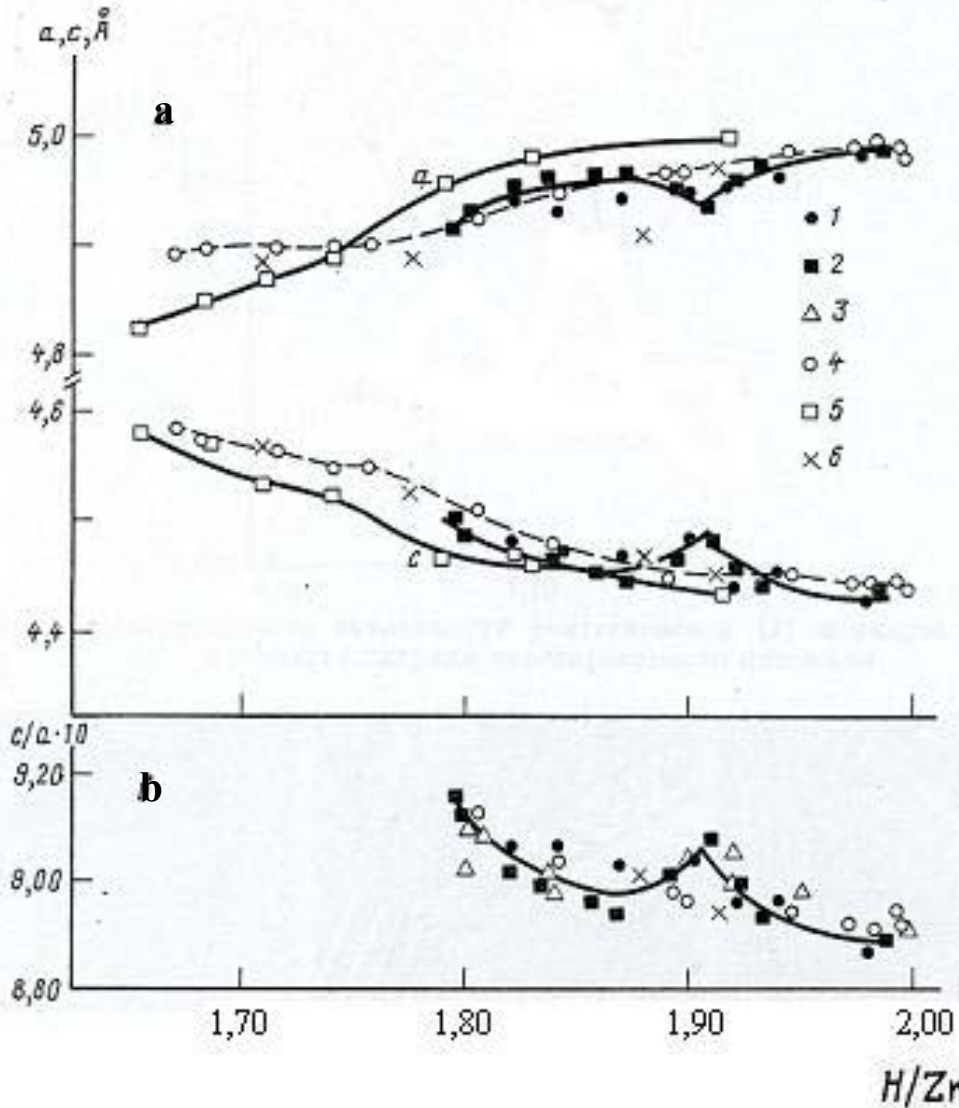


Time elapsed till measurement made after hydrogenation (month)

- o - 1,
- Δ - 6,
- \square - 12,
- \bullet - 24,
- \times - 48,
- \bullet - 60,
- \boxtimes - 24.

g/cm^3

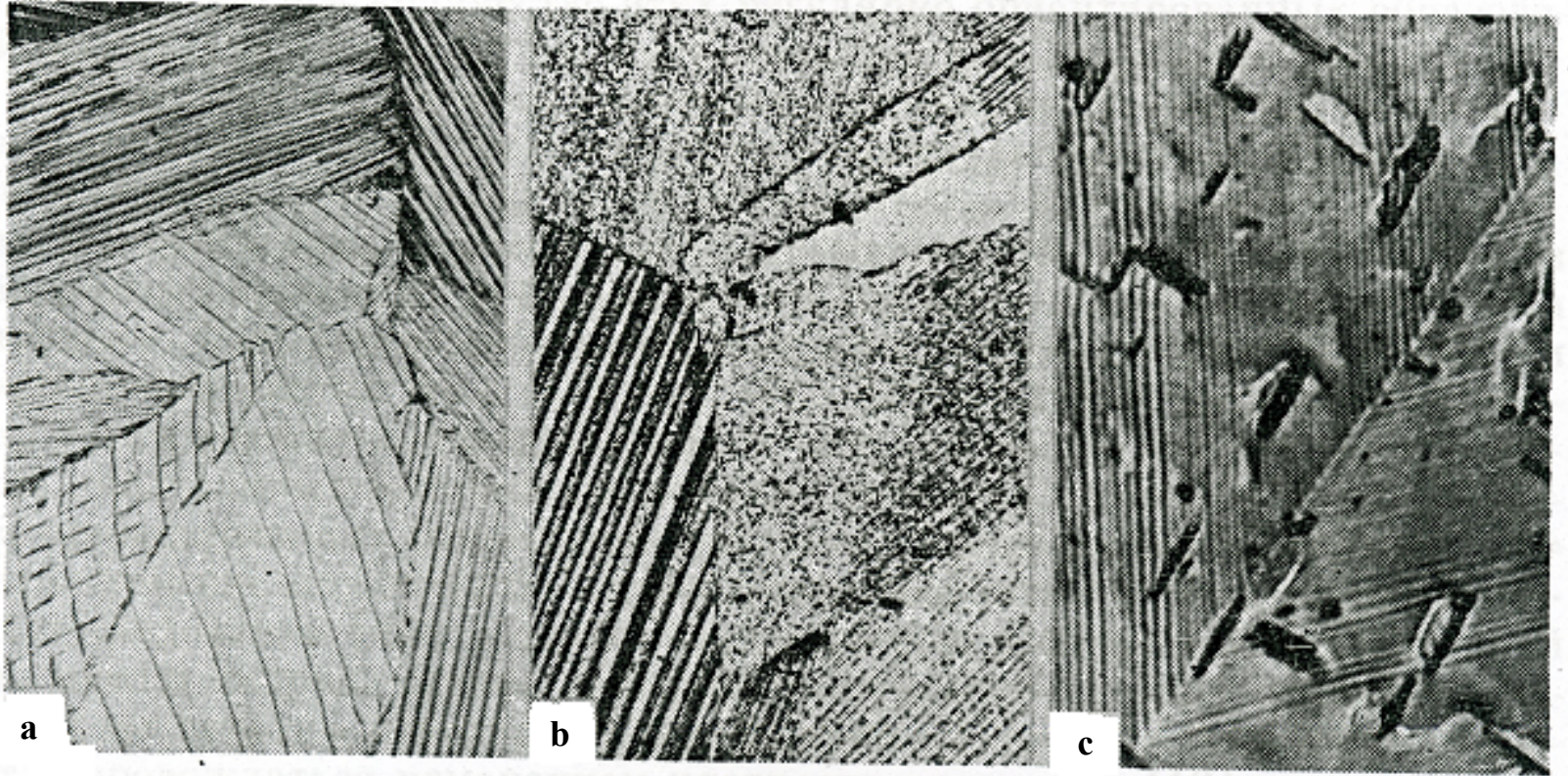
Concentration dependences of crystal lattice parameters (a); relationship c/a (b) of undoped zirconium hydride 3



Data from:

- [Syasin V. A. 1992],
- [--],
- [W. Mueller, 1968],
- [W. L. Korst, 1962],
- [R. Beck, 1962],
- [K. G. Barraclough, 1970].

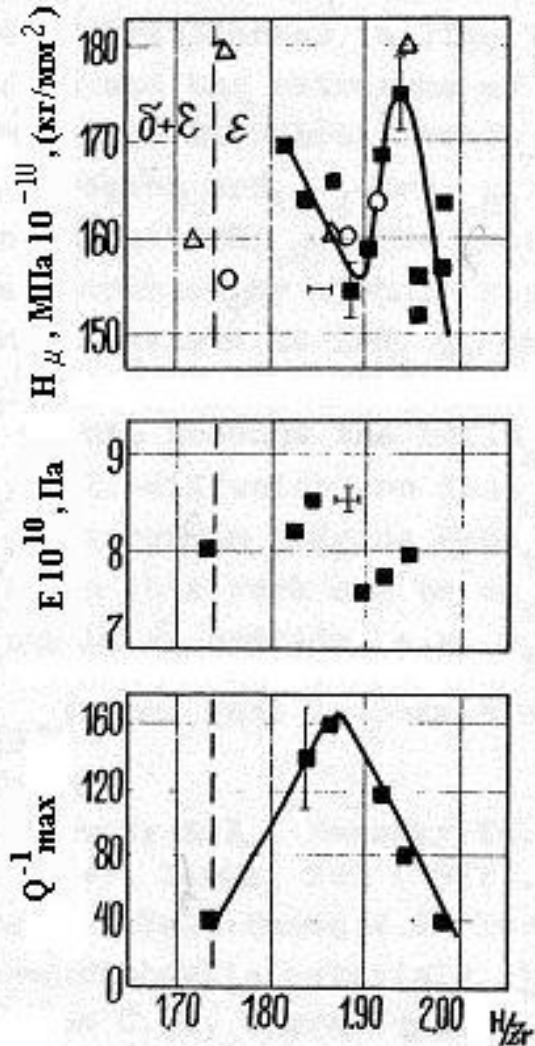
Structure of undoped zirconium hydride $ZrH_{1.94}$ after fabrication (a) $\times 100$ and storage during 10 years
(b) $\times 100$; (c) $\times 1300$ 4



Relationship between microhardness (H_{μ}), elasticity modulus (E) and internal friction (Q^{-1max})

of zirconium hydride composition **5a**

o - [K. G. Barraclough, 1970], Δ - [Andrievsky R. A, 1967]



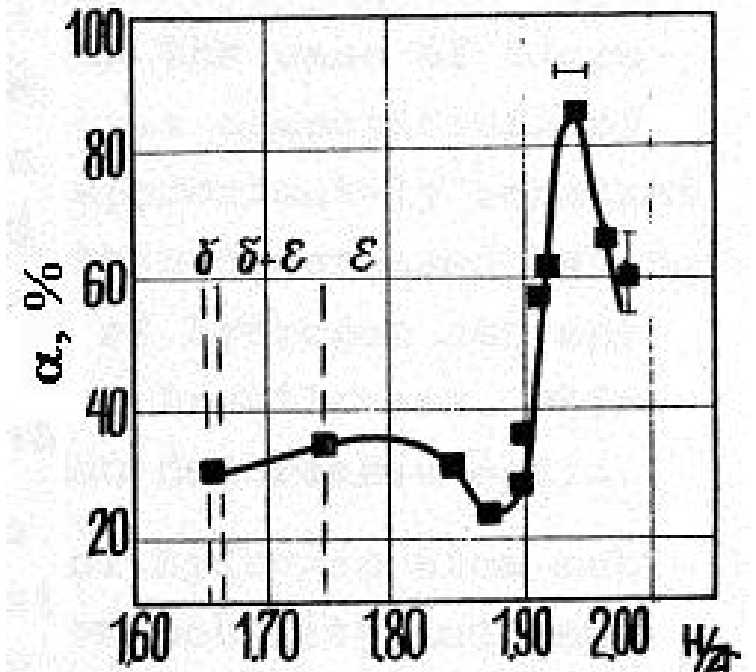
Zirconium hydride microbrittleness **5b**

$$\alpha = (N_c / N) \times 100\%$$

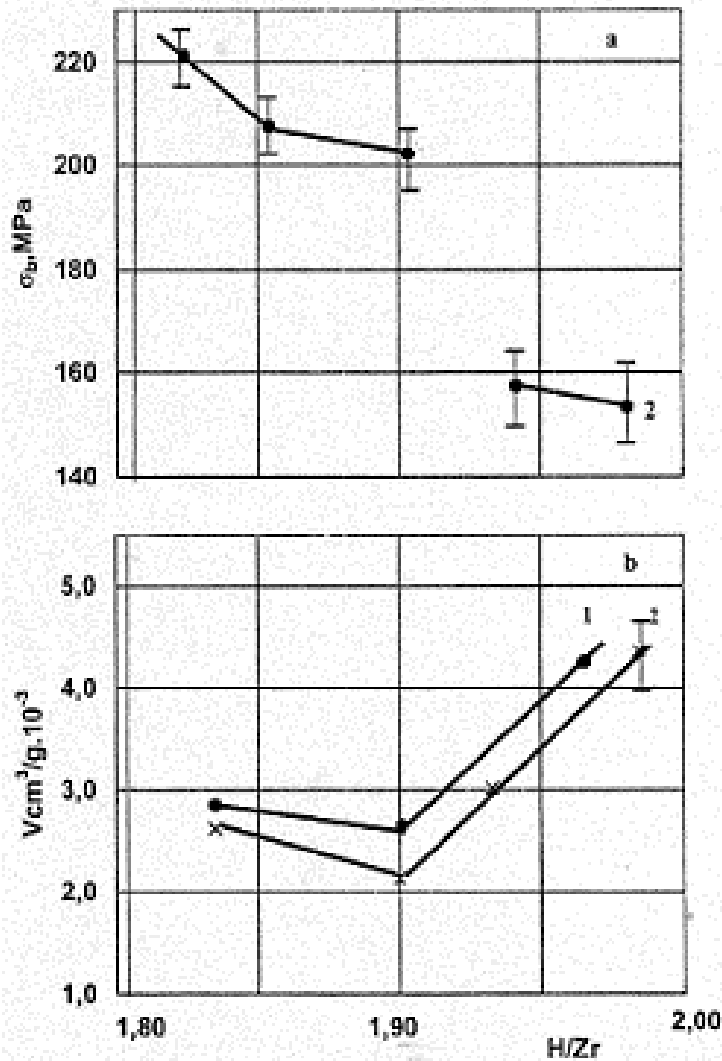
N – total number of indentor traces,

N_c – number of traces with cracks

Vickers diamond indentor

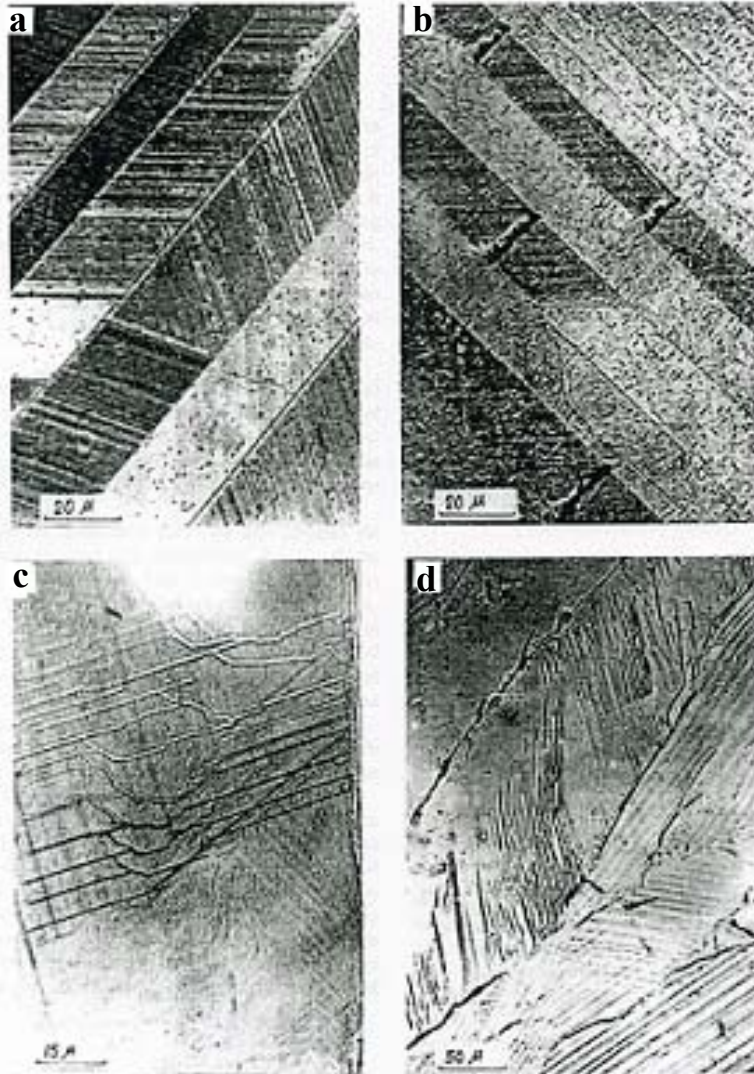


Changes in bending strength (σ_b) (a) and porosity (V) (b) ϵ -ZrH in relation to hydrogen content 6

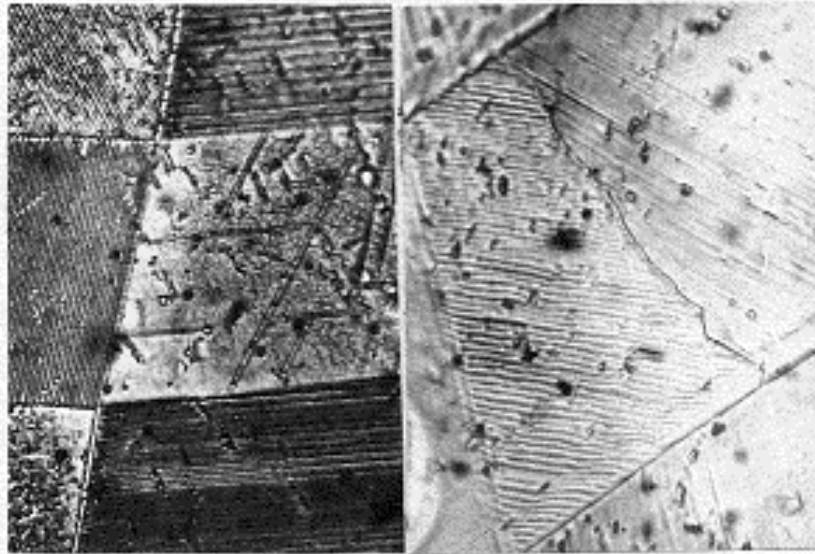


Grain size, μm
1 – (3000 – 5000),
2 – (40 – 80).

Structure changes and defect presence in zirconium ϵ -hydride depending on hydrogen content: (a) –twin structure $\text{ZrH}_{1.90} \times 800$, (b) –reorientation and microcracks in the twins $\text{ZrH}_{1.95} \times 800$, (c) – filling of channels - pores $\text{ZrH}_{1.95} \times 1000$, (d) – groups of channels – cavities in grains $\text{ZrH}_{1.98} \times 300$ 7



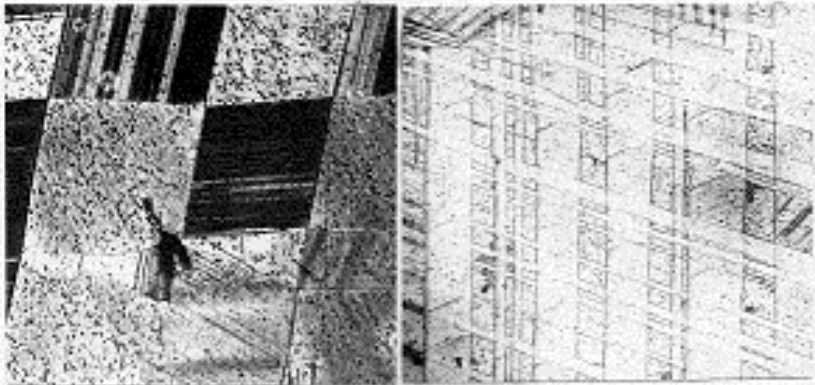
Structure changes and defect development in zirconium ϵ -hydride at twin interaction 8



a $\times 1000$

b $\times 2000$

- (a) – regions of twin intersection
- (b) – reorientation and development of cracks



c $\times 500$

d $\times 300$

- (c) – cracks in the region of twin collision
- (d) – multiple cracks in the region of twin intersection

Conclusions 9

1. Changes in density, cell unit parameters, hardness, brittleness, elasticity modulus of ϵ -zirconium hydride in the homogeneity region occur non-monotonically.
2. The extremums of concentration dependences for these characteristics correspond to $ZrH_{1.90}$ and $ZrH_{1.94}$ compositions.
3. For more than $ZrH_{1.90}$ compositions a sharp increase in the number of pores and microcracks is observed that causes a significant reduction of ϵ -zirconium hydride strength.
4. Presence of defects in ϵ -zirconium hydride in the form of pores, microcracks, channels is associated with peculiarities of formation of the twin structure and its further variation resulted from hydrogen saturation.

The authors express their thanks to Chernikov A. S., Vorobjev S. P., Khapov A. S. for their attention to the investigation and the discussion of the results.

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