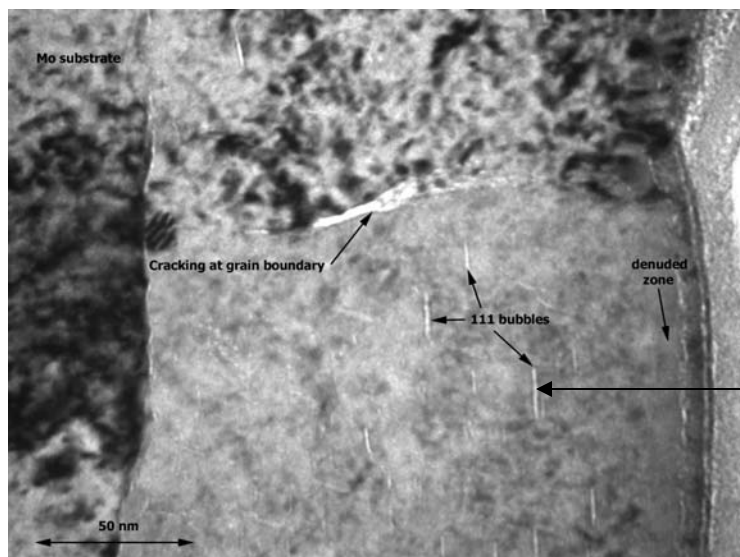


# Physics of He Platelets in Materials

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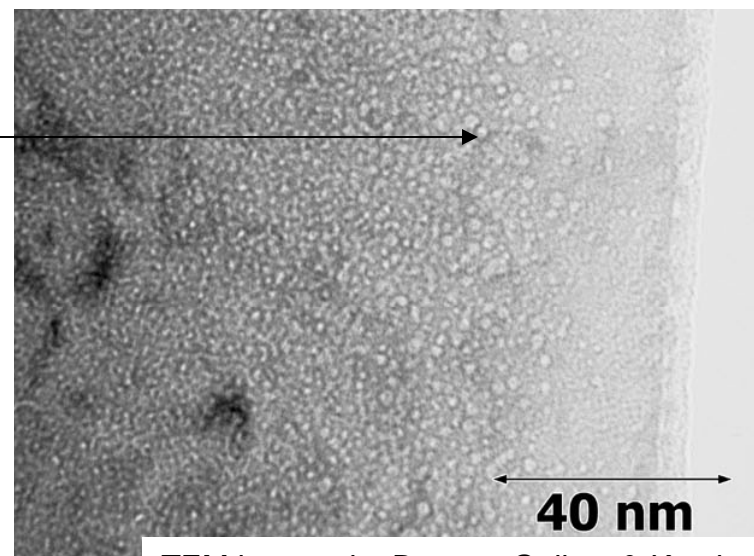
H and He in Materials Workshop, Albuquerque, April 14, 2005



*What causes the bubble shape difference?*

Pd tritide

Er tritide



TEM images by Brewer, Gelles, & Kotula

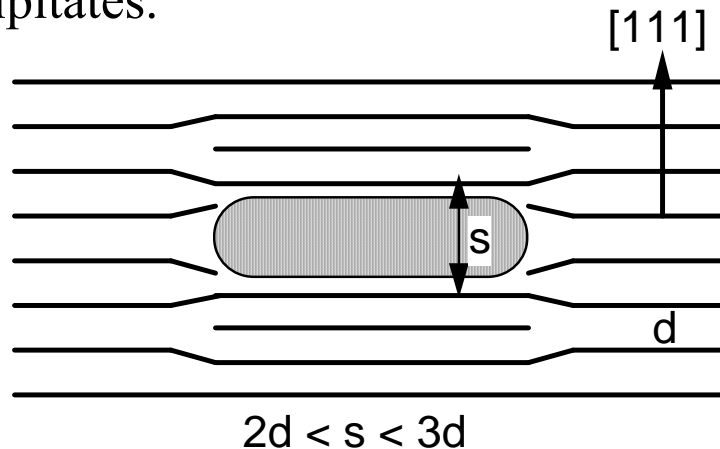
## Outline

- Platelet stability and growth
- Pd vs Er system
- Testing with XRD data
- Other materials & future efforts

*“In the spirit of a workshop,  
this is work in progress.”*

## Thin disk-shaped He platelets can exist with dislocation dipoles.

- In FCC structures, the  $[111]$  planes provide the greatest “space” for interstitial precipitates.

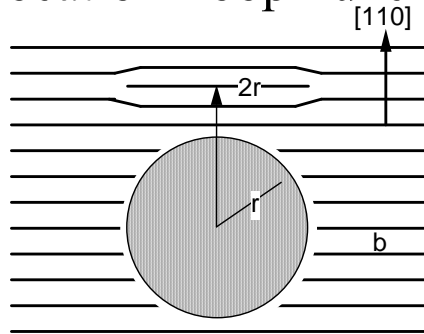


- Dislocations are formed when adjacent planes are displaced by  $d/2$ .
- The two dislocations remain attached to platelet provided adjacent  $(111)$  planes are concave toward platelet.
- The dislocations leave when the adjacent planes become straight.

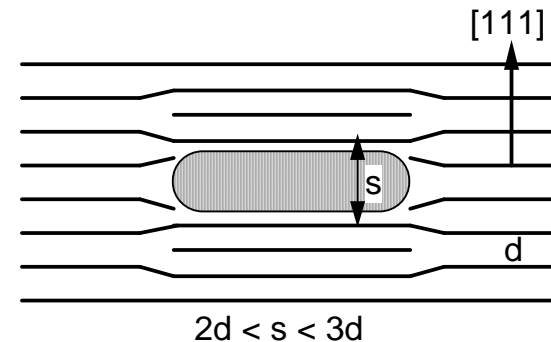
*The platelets are stable if they grow by radial expansion keeping  $s < 3d$ .*

# The pressure within the platelet has components due to surface and strain energies.

Dislocation Loop Punching:



Dislocation Dipole Expansion:



Surface Energy:  $p_e dV = \gamma dA$

$$p_e \pi r^2 b = \gamma 2\pi r b$$

$$p_e = 2\gamma/r$$

$$p_e \pi[(r+b)^2 - r^2] s = \gamma 2\pi[(r+b)^2 - r^2 + (r+b)s - rs]$$

$$p_e = (2\gamma/s) [(2r+b+s)/(2r+b)]$$

Lattice Strain: stress =  $\mu$  strain

$$p_s \pi r^2 = \mu [(b/2)/d] 2\pi r d$$

$$p_s = \mu b/r$$

$$p_s \pi[(r+b)^2 - r^2] = \mu [(d/2)/b] 2\pi[(r+b)-r]b$$

$$p_s = \mu d/(2r+b)$$

Bubble Pressure:  $p = p_e + p_s$

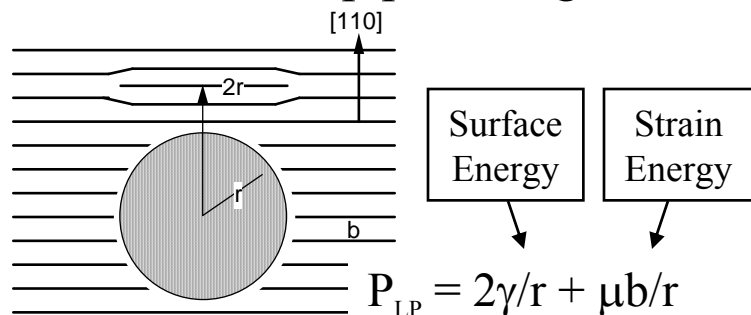
$$p_{lp} = 2\gamma/r + \mu b/r$$

$$p_{de} = (2\gamma/s) [(2r+b+s)/(2r+b)] + \mu d/(2r+b)$$

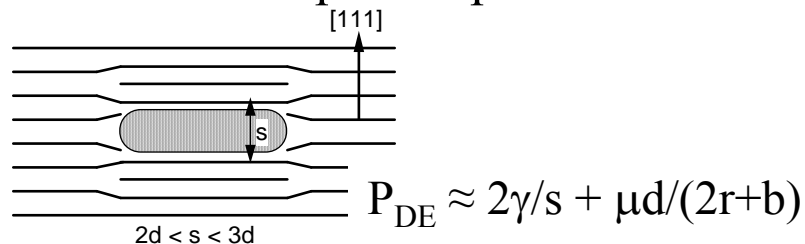
$$p_{de} = 2\gamma/s + \mu d/2r, \text{ at large } r$$

# The bubble shape and growth process depend on the tritide's mechanical properties.

- Dislocation loop punching



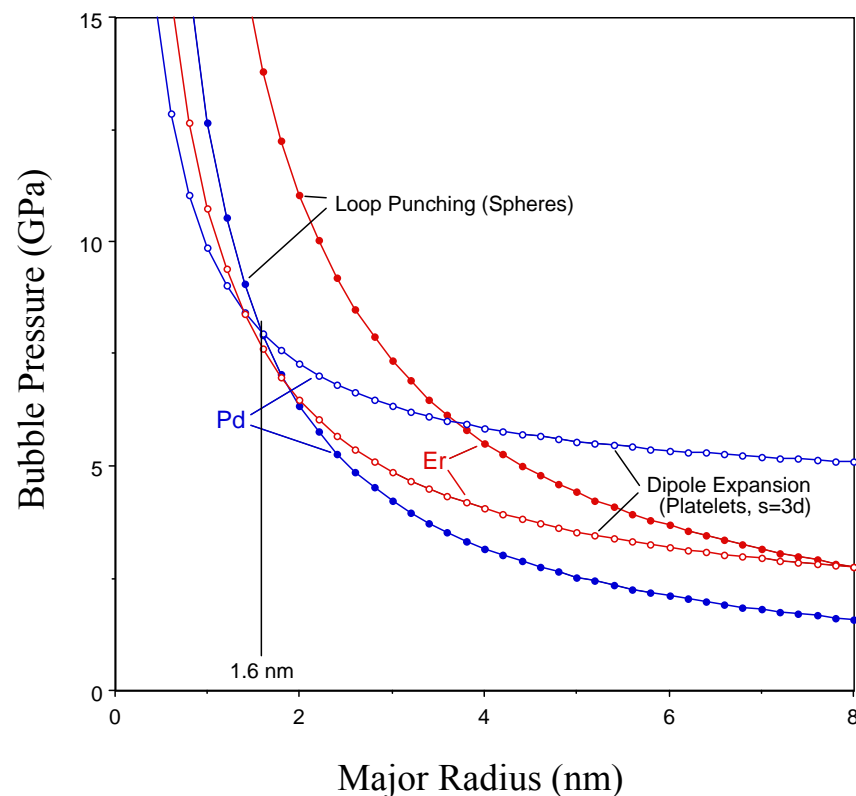
- Dislocation dipole expansion



- Thin, disk-shaped bubbles are caused by a low surface energy (low  $\gamma/\mu b$ ).

Note: Platelets are also the preferred shape in *young Pd tritide (<50 days)*.

Tritide	$\gamma$ (GPa-nm)	$\mu$ (GPa)	$b$ (nm)	$\gamma/\mu b$
Pd	1.54	33.6	.2852	0.16
Er	0.637	57.4	.3623	0.03



# This bubble shape condition holds for precipitates in a number materials.

Material	$\gamma$ (GPa-nm)	$\mu$ (GPa)	b(nm)	$2\gamma/\mu b$	shape
ErT	0.637	57.4	.3623 fcc	.061	platelets
ScT	0.954	54.0	.3382 fcc	.104	early platelets, then?
TiT	1.39	76.2	.3111 fcc	.117	platelets & elongated?
Ni	1.72	76.5	.2490 fcc	.180	spheres?
ZrT	1.48	32.6	.3522 fcc	.258	spheres
PdT	1.54	33.6	.2852 fcc	.322	spheres
Be	1.10	146	.359 hex	.051	platelets
Ti- $\alpha$	1.39	40.1	.291 hex	.238	platelets
W	2.22	158	.273 bcc	.103	platelets
V- $\alpha$	1.95	47.4	.263 bcc	.312	spheres
Nb- $\alpha$	1.90	38.2	.285 bcc	.350	spheres

- Surface energy/strain energy ratio for spherical bubbles
- Platelets are preferred for small r, where  $s/2r > 2\gamma/\mu b$ .

## Testing of pressure formulation is provided by lattice dilation data on aged tritides.

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- Tensile stress created by the precipitates produces a positive da/a.
- Spherical bubbles at Loop Punching pressure:

Hydrostatic tensile stress balances bubble pressure

$$p_{LP}(\Delta V/V) = B^*(3 da/a)_{LP}, \quad B^* = \text{bulk modulus of aged material}$$

where  $\Delta V/V = (4/3) \pi r^3 n_B$ ,  $n_B = \text{bubbles/cm}^3$

$$(da/a)_{LP} = p_{LP}(\Delta V/V) / 3B^*,$$

- Platelets at Dipole Expansion pressure:

[111] tensile stress balances platelet pressure, 4 components

Projection along [100] cubic axes =  $1/\sqrt{3}$

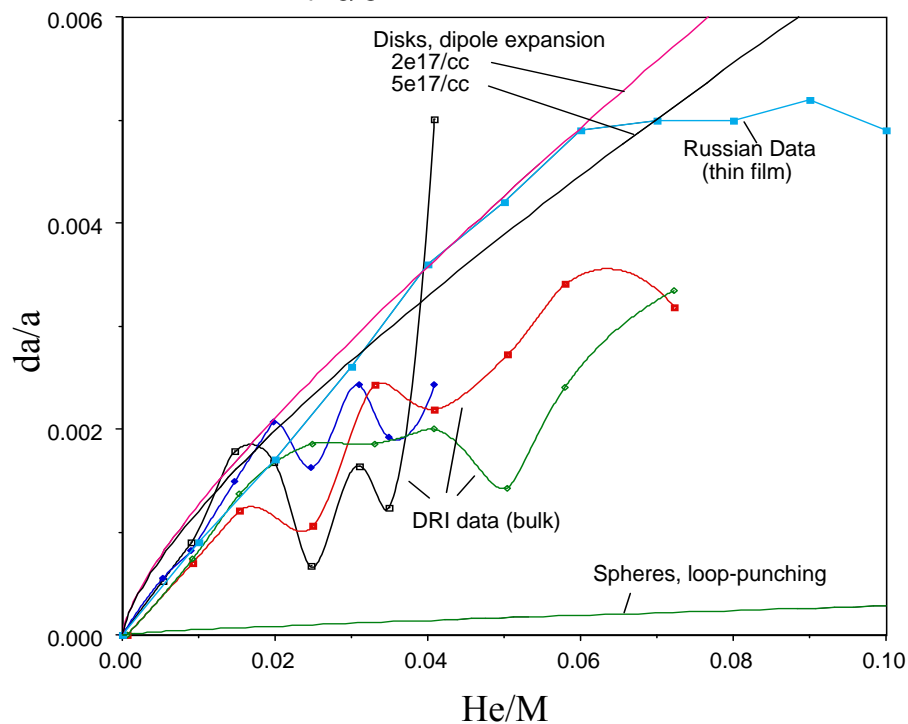
$$p_{DE}(\Delta A/A) 4/\sqrt{3} = E^*(da/a)_{DE}, \quad E^* = \text{Young's modulus of aged material}$$

where  $\Delta A/A = \pi r^2 (n_B/4)^{2/3}$ ,  $n_B = \text{bubbles/cm}^3$

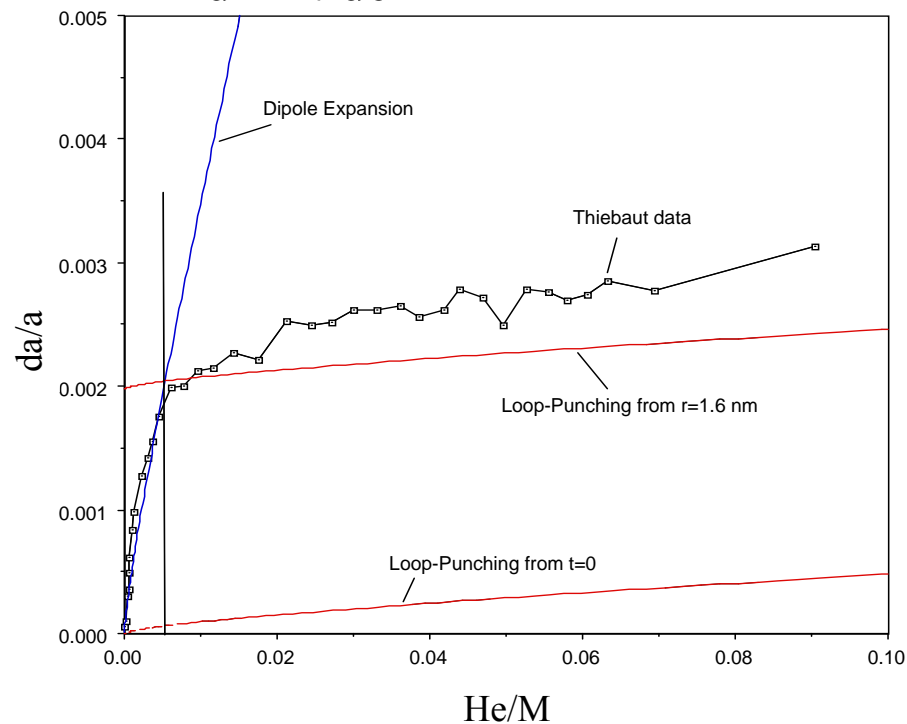
$$(da/a)_{DE} = 4 p_{DE}(\Delta A/A) / E^*\sqrt{3}$$

# Lattice dilation data for both ErT and PdT show the effects of rapid platelet growth.

## Er Tritide



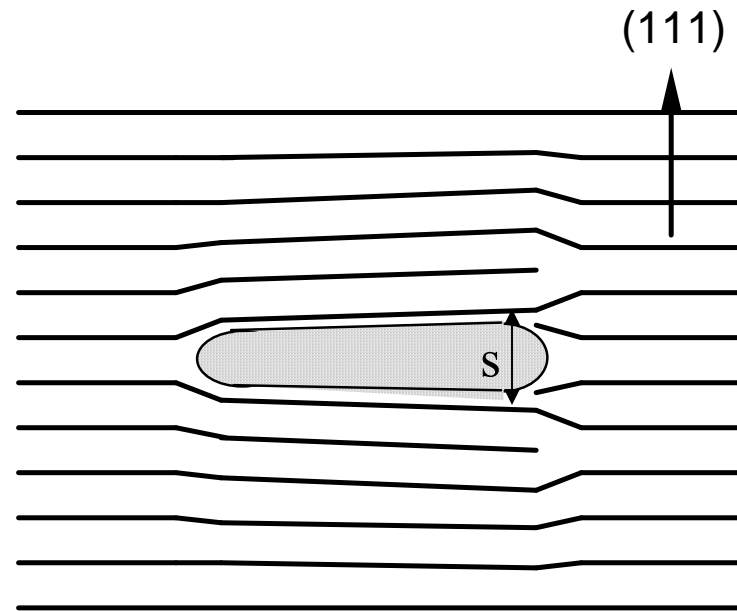
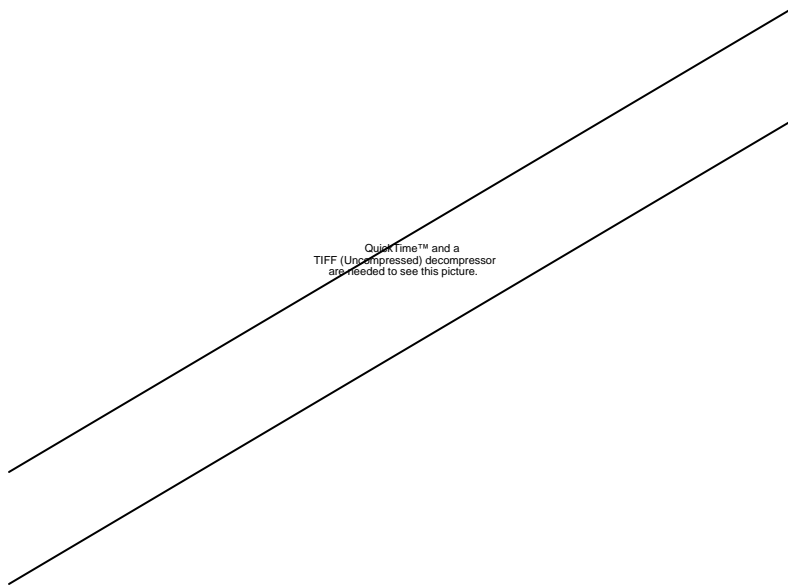
## Pd Tritide



- Neither can be explained by dislocation loop punching alone.
- The model produces the correct transition point for the Pd system.
- How are platelets formed?

# Transition of a nano-crack to a platelet can begin with a dislocation near the crack tip.

PdT<sub>0.6</sub> aged 3 months



$$s \approx 2d = 0.47 \text{ nm}$$

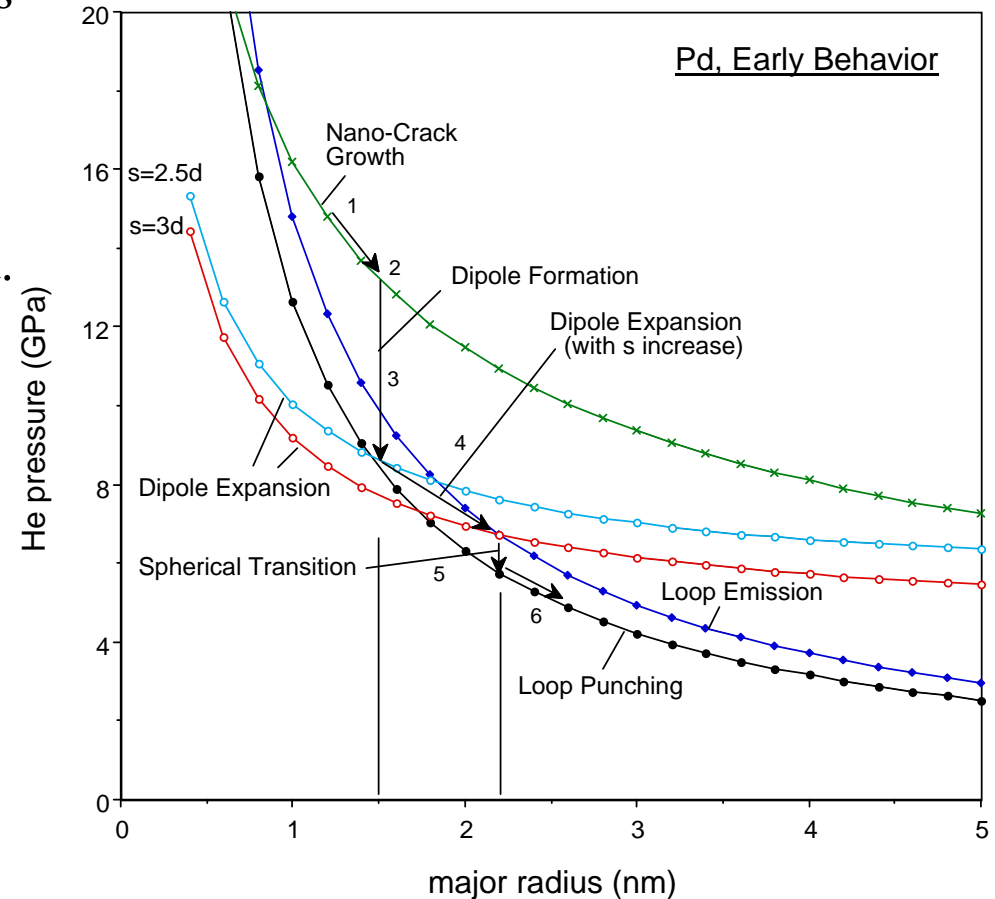
HR-TEM [110] view (Thiebaut)



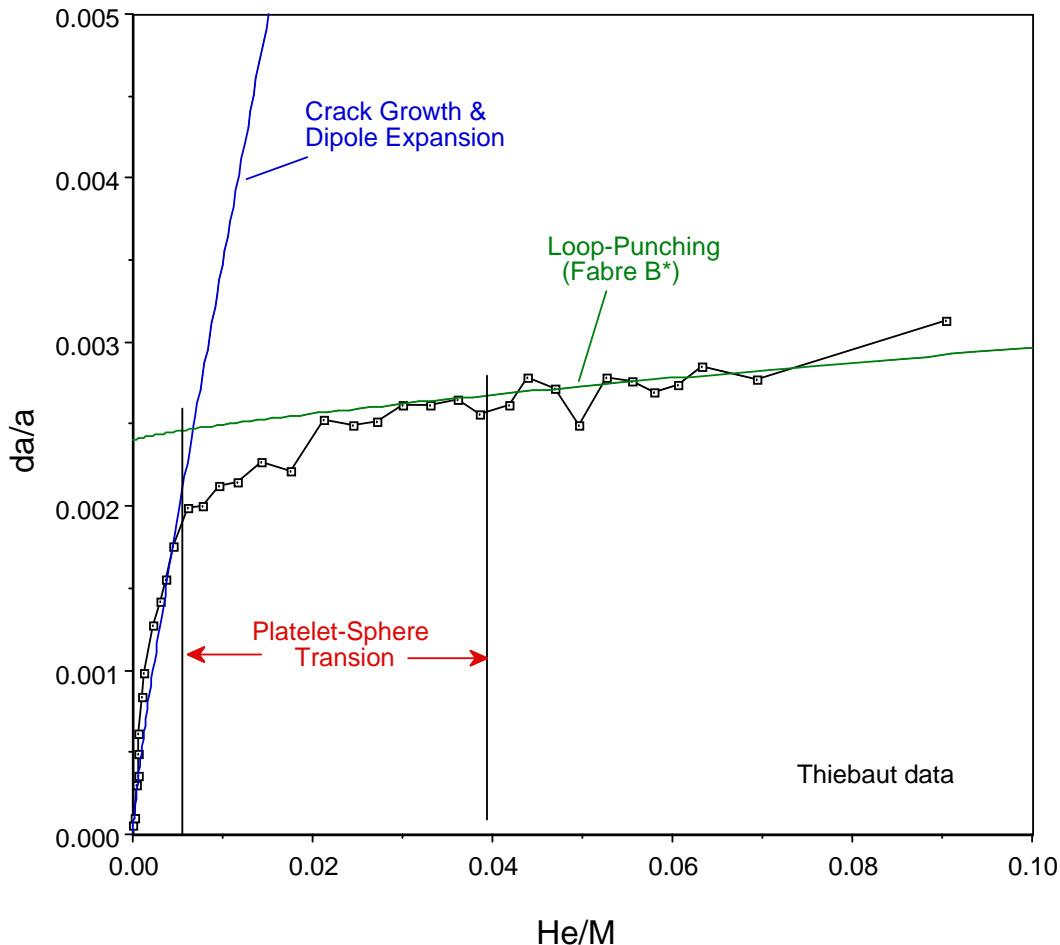
# The early growth of bubbles in PdT appears to have several stages.

1. He atoms collect in (111) planes and open nano-cracks (Griffith):  

$$P_{nC} = 4\gamma/s, \quad s = 4[\gamma(1-\nu)r/\pi\mu]^{1/2}.$$
2. Dislocation dipoles form when the nano-crack gap reaches  $s=2d$ .
3. Platelet pressures drop as their thicknesses increase to  $s \approx 2.5d$ .
4. The platelets expand radially until  $s=3d$ , where the dipole escapes.
5. [110] loops are emitted as the platelets transition to spheres.
6. Spherical bubbles continue to grow by normal loop-punching.

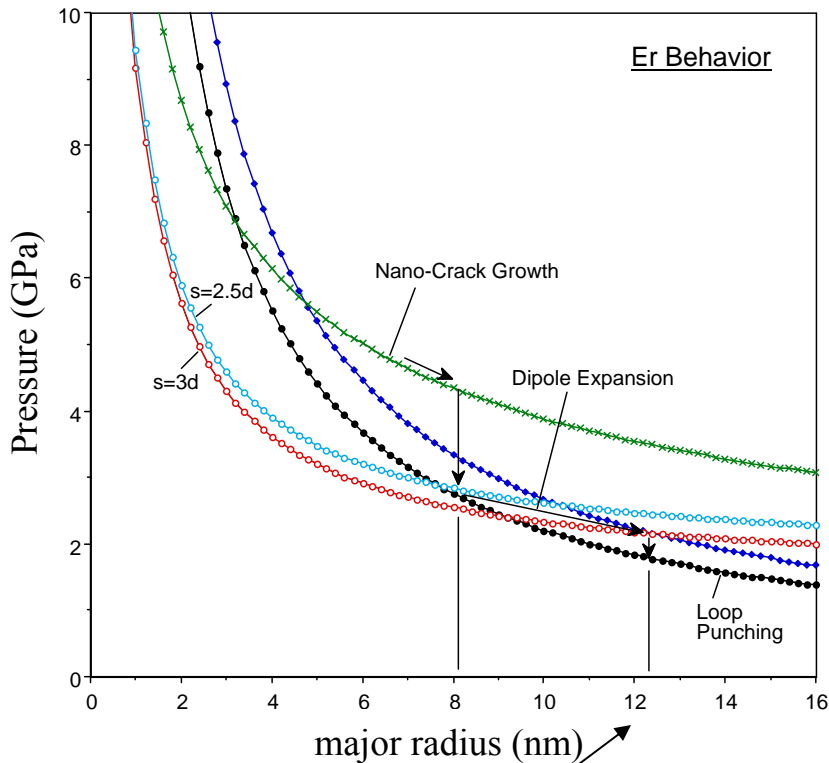


# Lattice dilation “details” of PdT appear to support the existence of multiple stages.



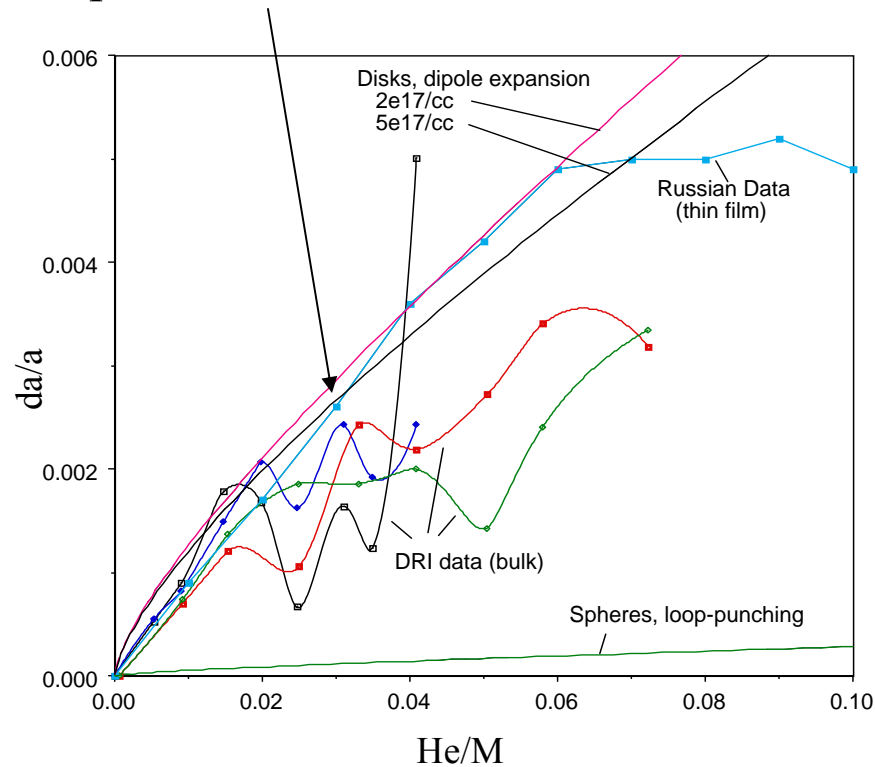
- Initial Griffith crack growth at high pressure will produce an even rise.
- The bubble volume increases by 8X during the transition from platelets to spheres.
- Emitted dislocations must remain trapped between “bubbles”.
  - Bubble source volumes remain constant!)

# The He precipitates in Er tritide remain 2-dimensional throughout life.



- The linking of large platelets by should begin prior to the spherical transition, where  $\Delta A/A \approx 3$ .

- Pressure changes during transions from nano-cracks to platelets may explain oscillations in bulk da/a data.



## Work on platelet structures is continuing.

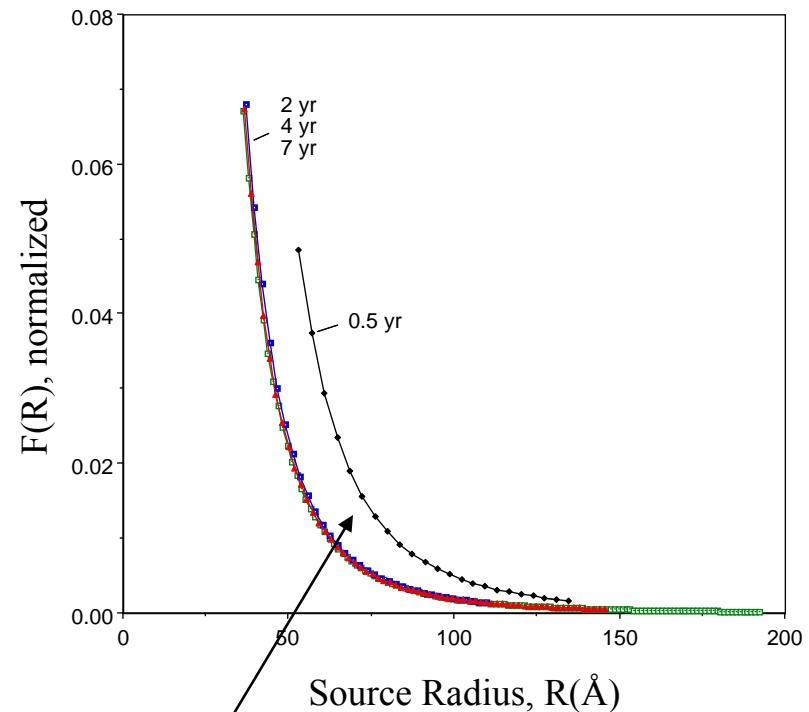
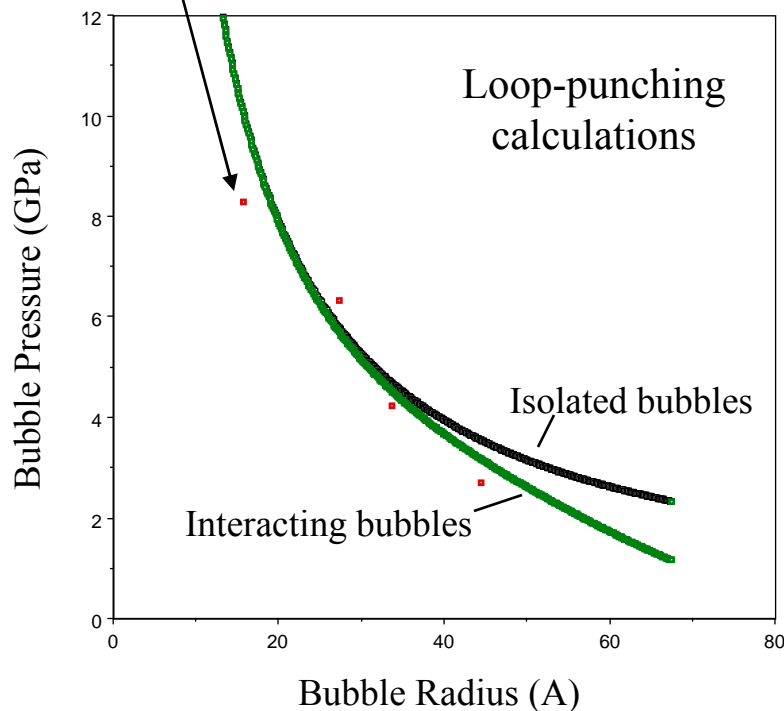
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- Thin (111) platelet bubbles can be associated with nano-cracks or dislocation dipole structures.
- Additional theoretical work is examining
  - formulations of platelet characteristics
  - linking of platelets by inter-platelet fracture
- Continued testing will examine
  - the bubble pressure and spacing distribution from 6 month-old Pd tritide (in spherical transition stage).
  - observed bubble shapes in other materials (e.g. SiC).
  - observed bubble shapes in implanted materials.
  - early decrease of Pd tritide plateau pressure (better data needed).
  - proposed TEM studies in selected materials to characterize shapes and identify transition points.
- A comprehensive effort to obtain model-testing data should improve early bubble characterization in PdT.



# The bubble pressure and spacing distribution for young PdT deviate from spherical growth.

- A platelet shape will lower the computed bubble pressure at 0.5 yr  
- from 10 to 8 GPa,  
and produce better agreement with the pressure deduced by NMR.



- It will also change the bubble spacing distribution computed for the 0.5 yr sample.

Technique	Information	Existing Pd Data ( <a href="#">agree</a> , <a href="#">disagree</a> )
TEM: Brewer, Gelles (PNNL)	He bubbles <u>Number density</u> , size, distributions Profiles, effect of intrinsic defects on nucl. <u>Onset of IB-fracture, <math>\mu</math>-fracture network percol.</u>	$n_B$ - Thomas, $n_B$ - Thiebaut $r(\text{He}/M)$ - Thiebaut
NMR: Curro (LANL), Cowgill	He atoms in bubbles He density distrib. (& pressure from EOS) Changes with temp, anneal, phase <u>Effect of bubble interactions / IB-fracture</u>	$v_{\text{He}}$ (dist, He/M) - Abell $\Delta v_{\text{He}}/\Delta\Phi$ - Cowgill
Swelling:	Integrated bubble volume, $\Delta T$ & $\Delta\Phi$ effects (Onset of IB-fracture?)	$\Delta V/V(\text{He}/M)$ - Guthrie, Abell
XRD, PCT: Majzoub, Walters (SRNL)	Lattice strain, mean bubble press., $\Delta T$ effects PCT shifts (plateau, boundaries)	$da/a$ - Thiebaut $\Delta p_H$ - Guthrie
Early He Release (ERF): Cowgill, Mintz (LLNL)	He escape depth, bubble profile <u>Dependence on nucleation conditions</u>	ERF - Meyer
He Thermodesorption: Walters (SRNL)	<u><math>\mu</math>-fracture network percolation</u> <u>Effects of powder particle size</u>	Crit. He/M - Coronado, Meyer
He-Implanted Material: Hertz	TEM, NMR, swelling for growth & fracture data; XRD for lattice strain; n-indent for $E^*$	$\Delta V/V(\text{He}/M)$ - Pontau, Hertz $n_B$ - Gelles
Nano-indentation: Moody, Knapp	Mechanical props. of un-aged & <u>aged material</u> <u>Elastic modulus of composite</u>	$c_{ij}(t_0)$ - (Wolfer), $c_{ij}(t)$ - Schwarz $E^*(t_0)$ - Fabre
He Implant/Re-emission: Cowgill	He diffusivity, <u>pairing energy</u> <u>Temperature dependences</u>	$D_{\text{He}}$ - Cowgill
SAXS/SANS/PAS/EXAFS/EELS: Majzoub, Browning, Hertz	Bubble data? (Onset of bubble coalescence / IB-fracture?)	R(EXAFS) - Thiebaut

