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U.S. Department
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New phases in the good old manganites

or

*Structural trends in oxygen-vacancy-ordered
 $\text{La}_x\text{Sr}_{1-x}\text{MnO}_y$ perovskite manganites and the
 $\text{A}_{4+n}\text{B}_{4+n}\text{O}_{10+3n}$ homologous series.*

Leopoldo Suescun

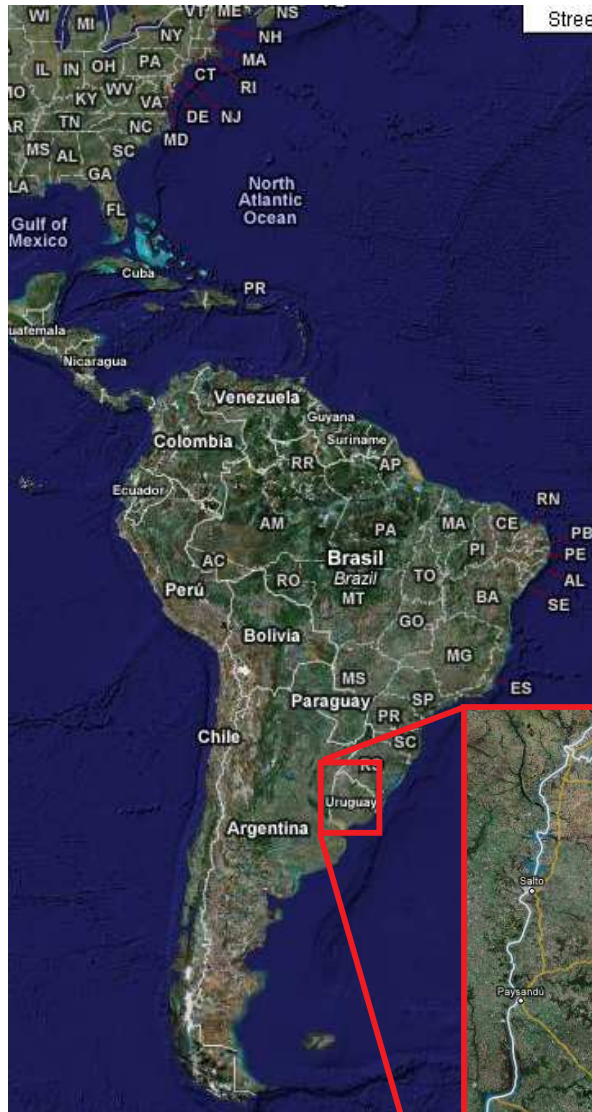
*Neutron and X-ray Scattering Group - Materials Science Division
Argonne National Laboratory, Argonne, IL, USA*

*Materials Design Lab. – Physics Department
Northern Illinois University, DeKalb, IL, USA*

&

*Cryssmat-Lab/DETEMA – Facultad de Química
Universidad de la República – Montevideo – Uruguay.*

A bit of personal history



- BSc (1995), MSc (1999) and PhD (2003) in Chemistry (Crystallography) at Universidad de la República (Uruguay State University) in Montevideo, Uruguay. Thesis Advisor: Prof. Alvaro W. Mombrú
- Assistant Professor of Crystallography (with tenure) since 2001, on paid leave.



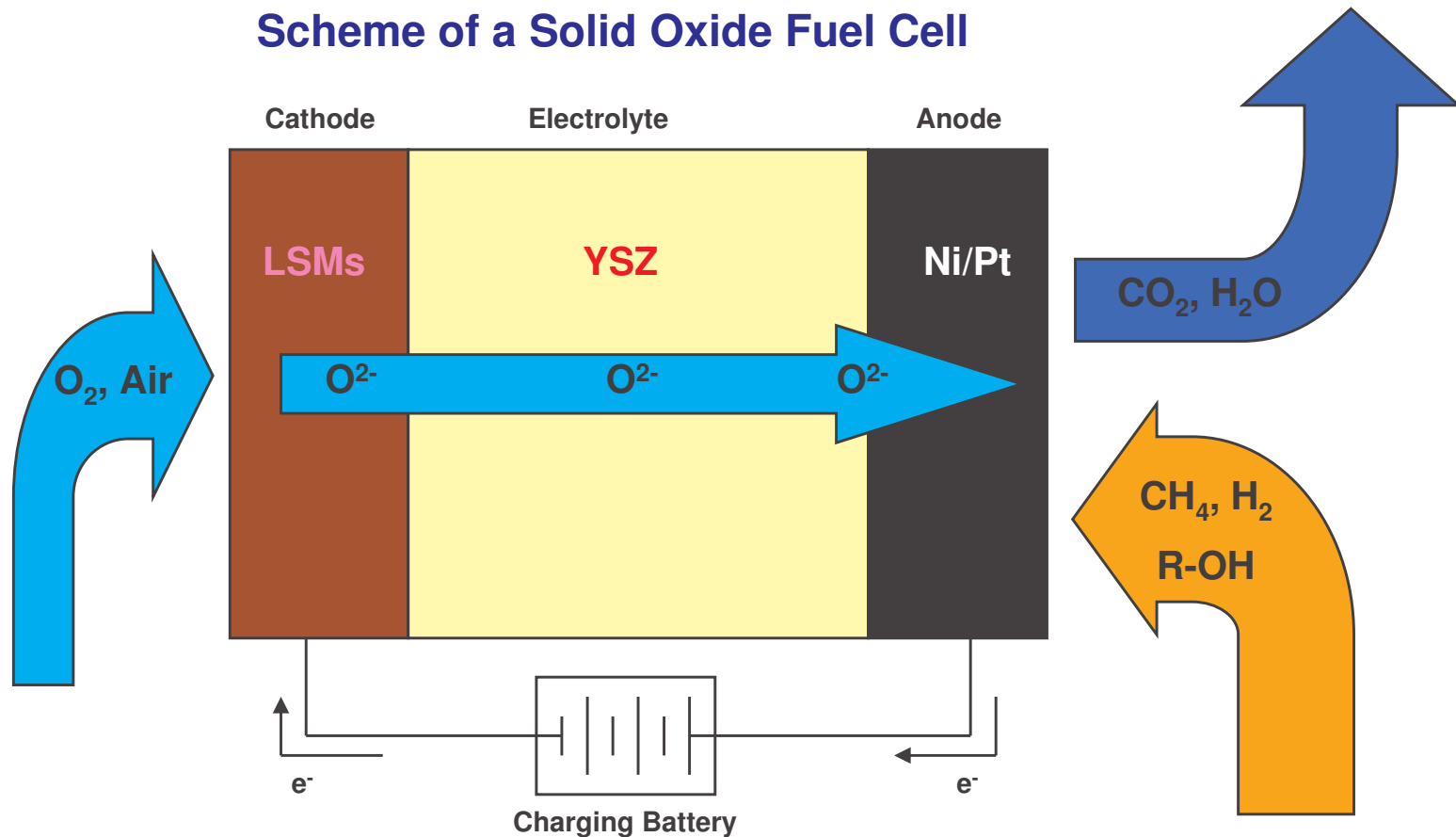
- Postdoctoral Appointee, Materials Science Division, Neutron and X-ray scattering Group.
 - Since April 29th, 2005
 - Leaving on April 28th

Overview

- Motivation: Perovskite materials for S.O.F.C. cathodes
- Experimental work and initial results: NPD experiments in the $\text{La}_x\text{Sr}_{1-x}\text{MnO}_y$ system.
- Re-focusing: Oxygen-vacancy, charge and orbital ordering in $\text{Sr}_5\text{Mn}_5\text{O}_{13}$ and $\text{Sr}_7\text{Mn}_7\text{O}_{19}$
- A new homologous series $\text{A}_{4+n}\text{B}_{4+n}\text{O}_{10+3n}$ built upon BO_5 pyramids and BO_6 octahedra
- Conclusions, take home message.
- Perspectives

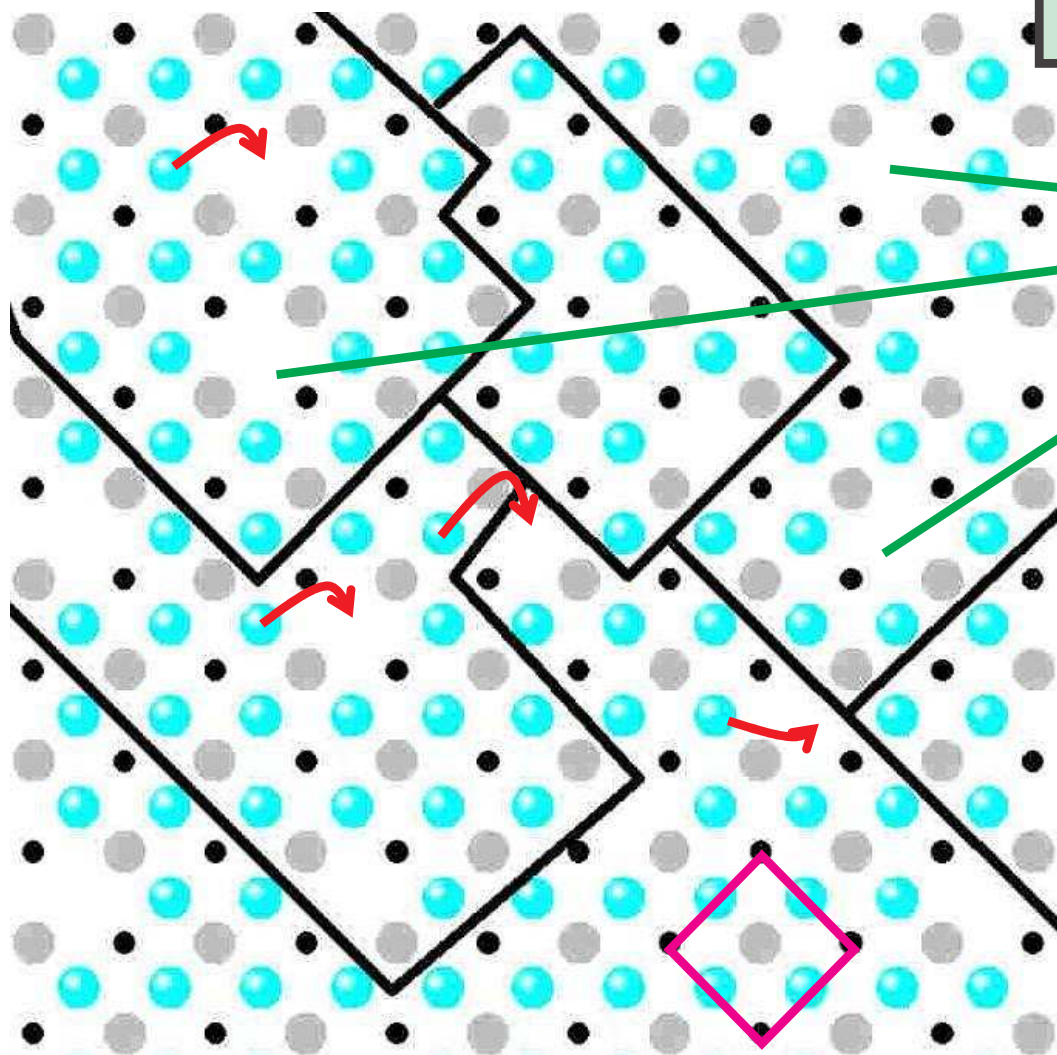
Solid Oxide Fuel Cell (SOFC):

- Device for efficient conversion of chemical energy into electricity.



- SOFC cathode materials: *fast oxygen conductivity & electronic conductivity.*

Effects of oxygen/vacancy ordering in cathode materials for SOFC:



Disordered oxygen vacancies

O^{2-} moving through low energy barriers

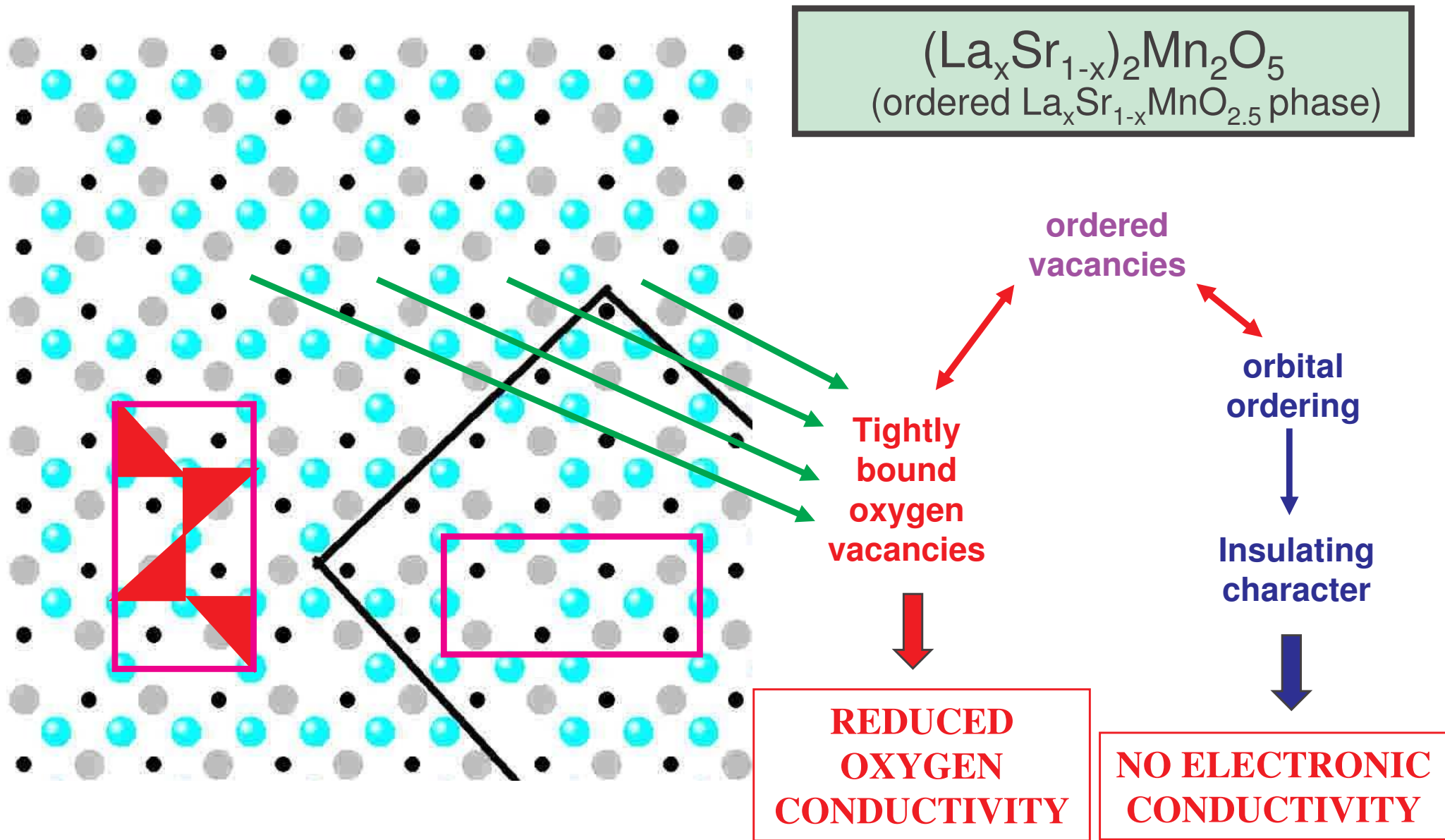
orbital disorder

Metallic character

FAST OXYGEN CONDUCTIVITY

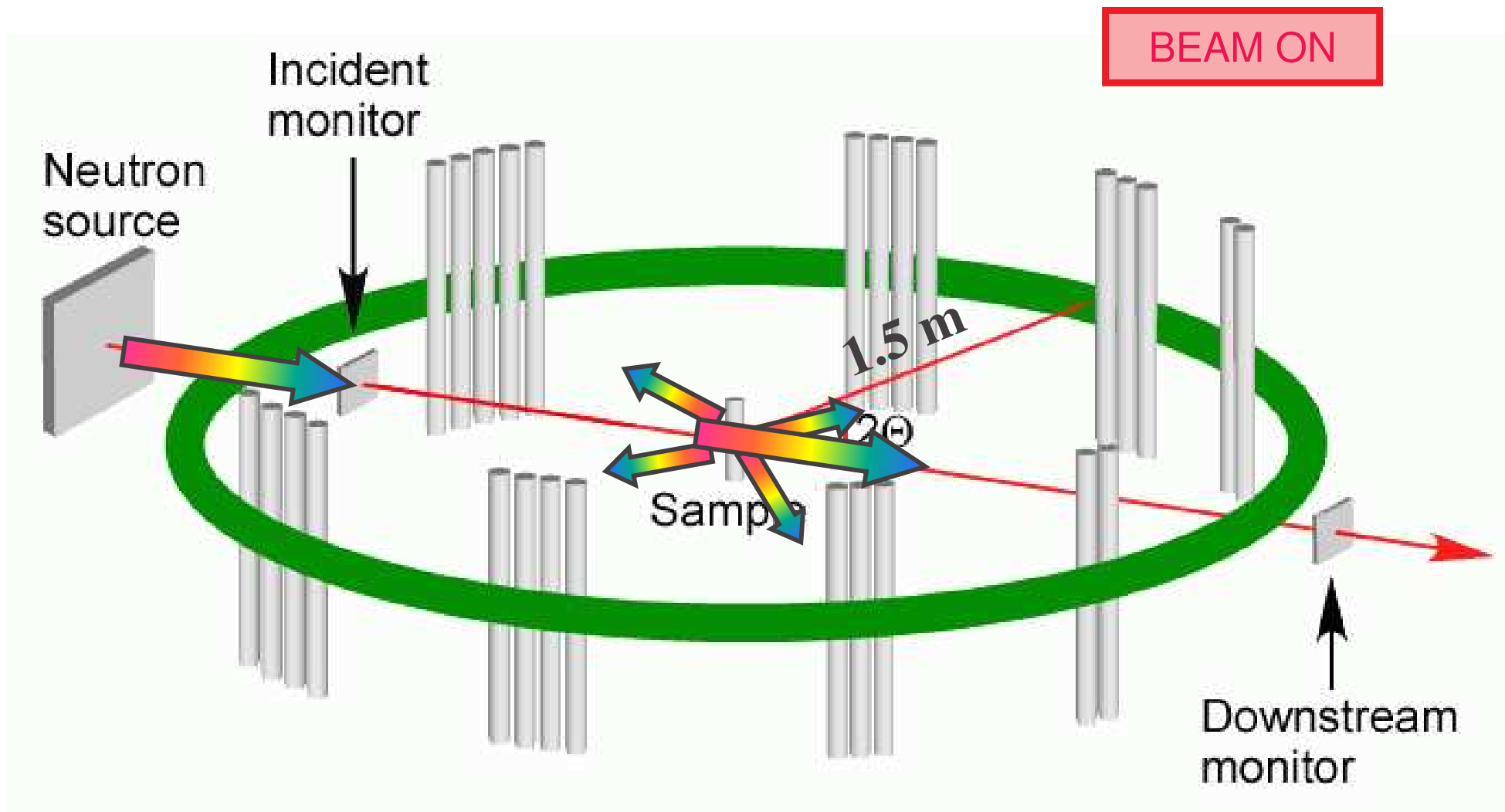
ELECTRONIC CONDUCTIVITY

Effects of oxygen/vacancy ordering in cathode materials for SOFC:

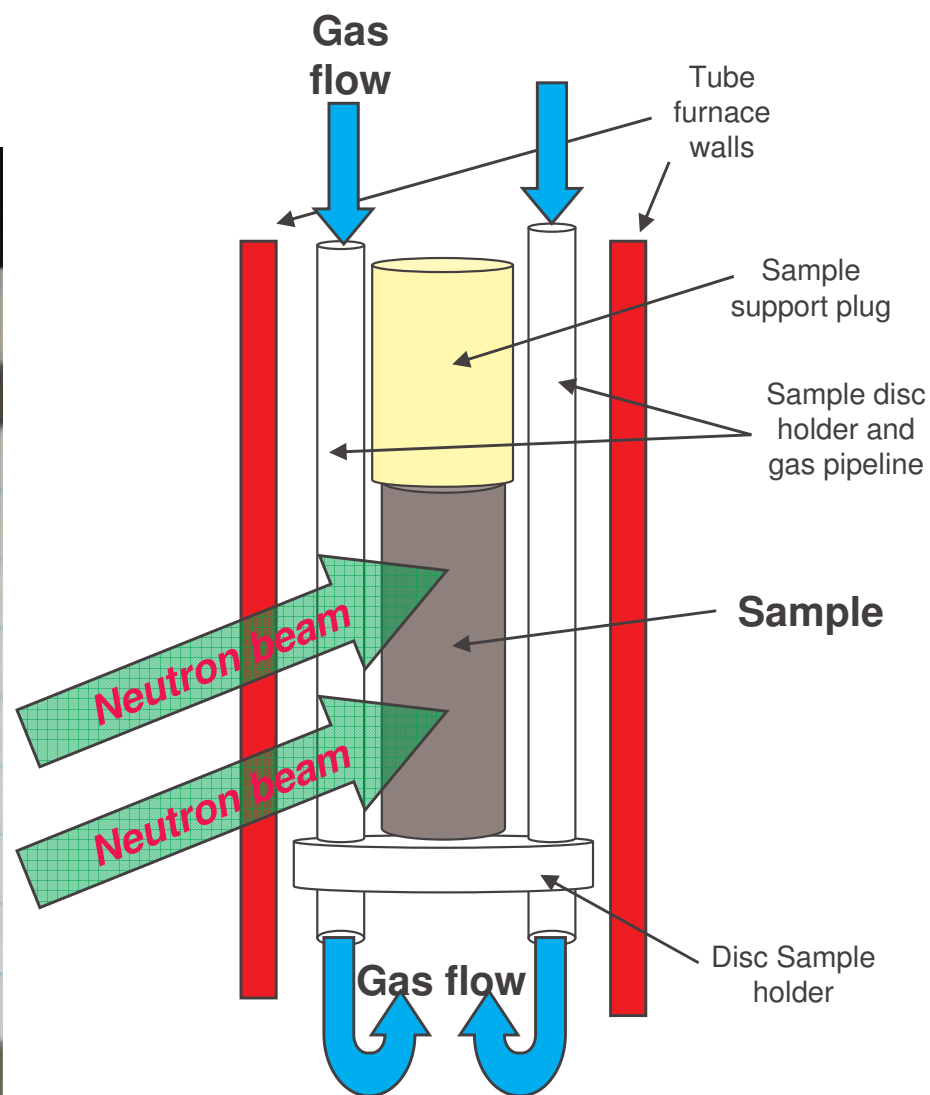
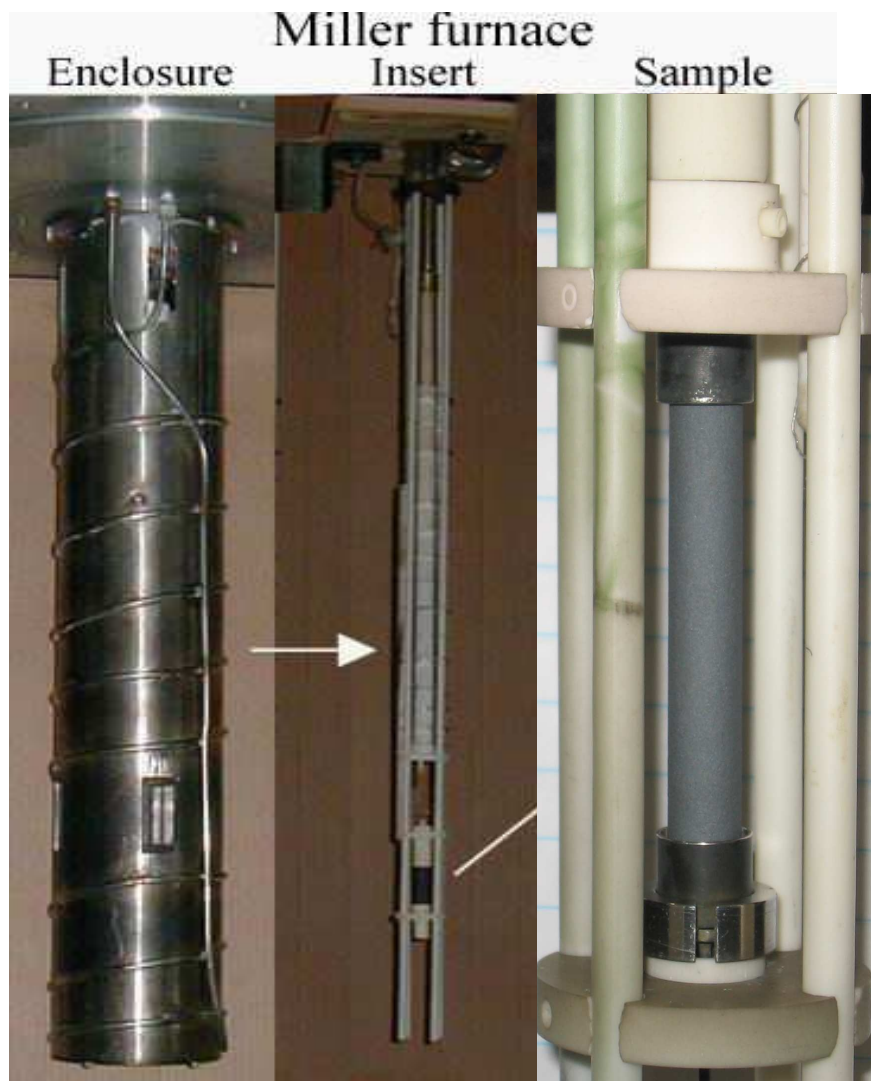


TOF Diffractometer Setup at former IPNS facility

Schematic Illustration of SEPD/GPPD



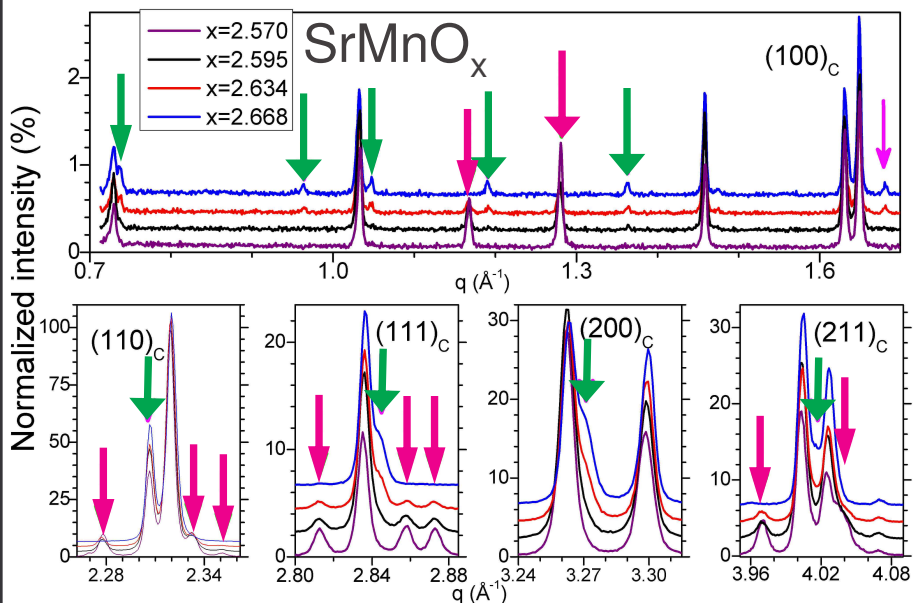
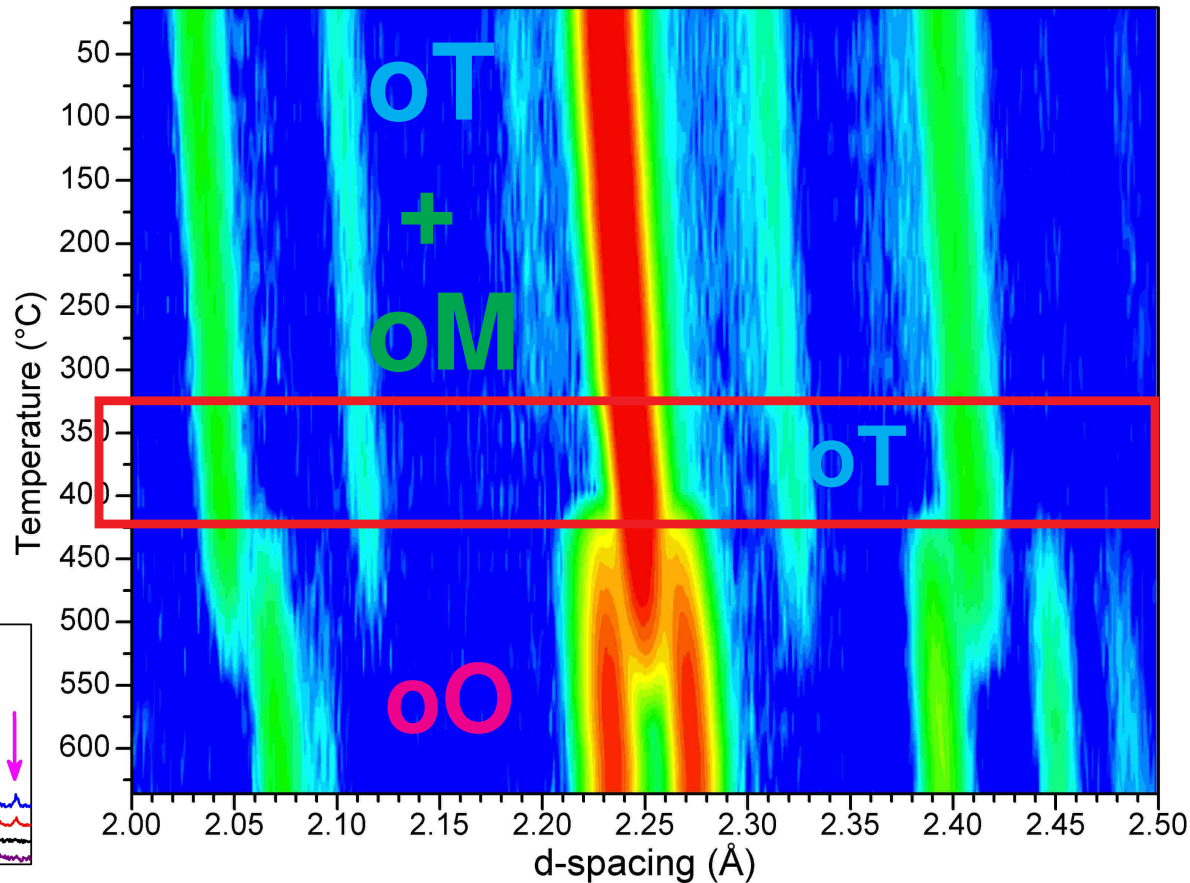
In-situ neutron diffraction experiment setup



From RT to 1200°C in O₂, Air, Ar, CO, CO₂, mixtures, etc. and var. flow rate.

SrMnO_x: In-situ NPD experiment

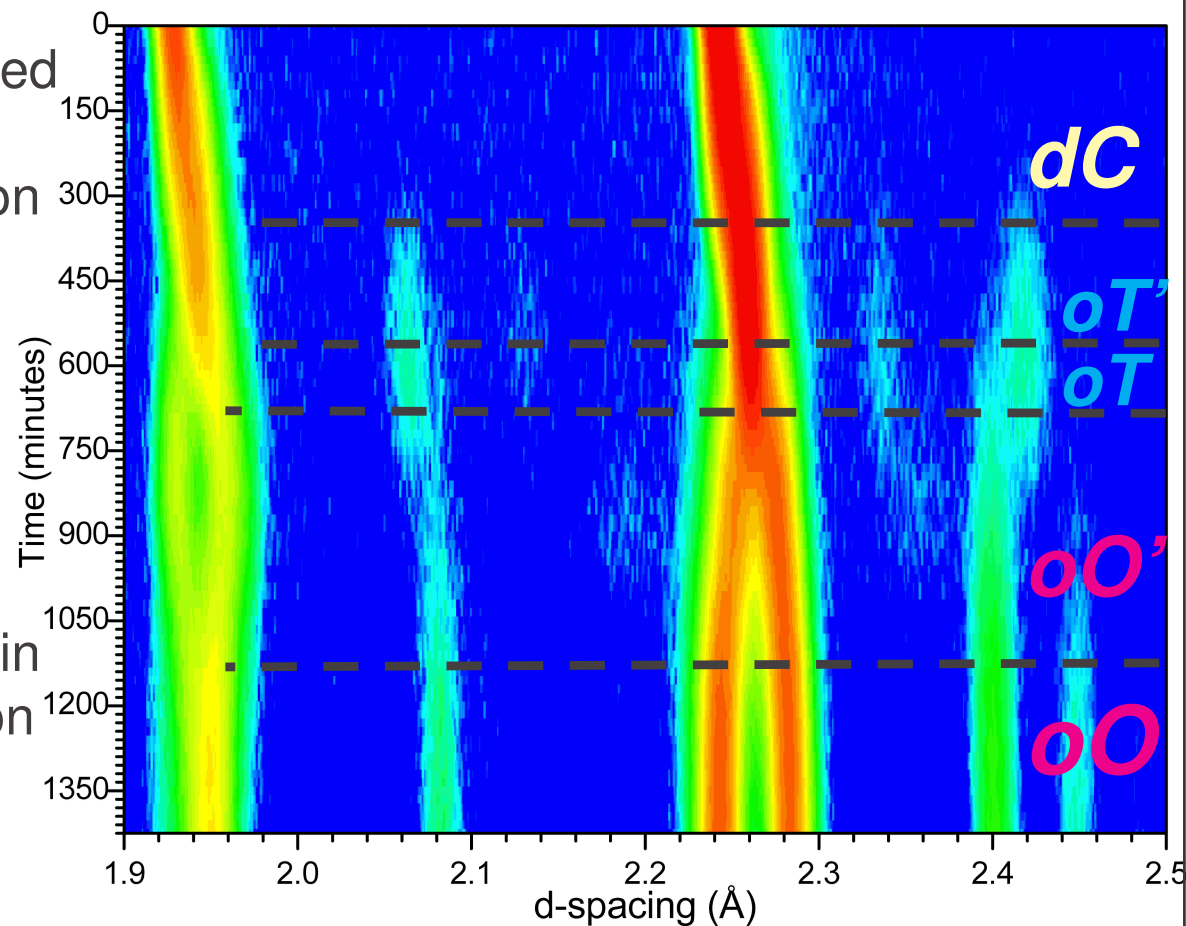
- Initial Interpretation: Starting sample loses oxygen and a phase transition occurs between vacancy ordered phases $\circ\text{T-Sr}_5\text{Mn}_5\text{O}_{13}$ and $\circ\text{O-Sr}_2\text{Mn}_2\text{O}_5$.



- Reality was more complex and required high-resolution X-ray diffraction (APS): Initial sample was two-phase, one of them never reported. The new phase $\circ\text{M-Sr}_7\text{Mn}_7\text{O}_{19}$ converts to $\text{Sr}_5\text{Mn}_5\text{O}_{13}$ just before $\text{Sr}_2\text{Mn}_2\text{O}_5$ starts forming.

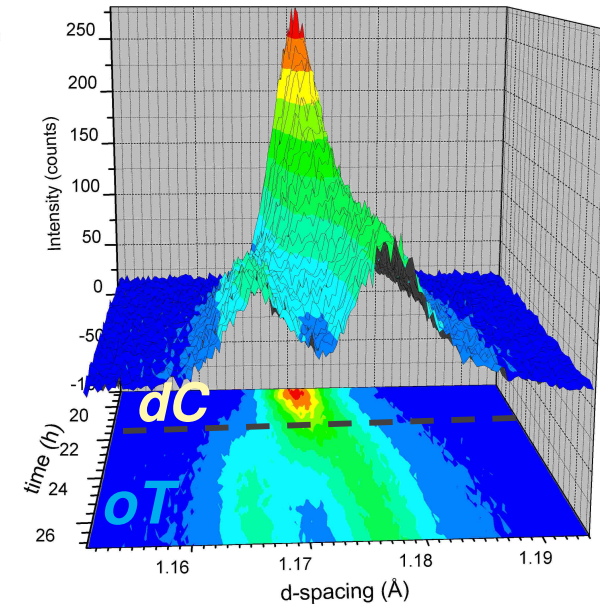
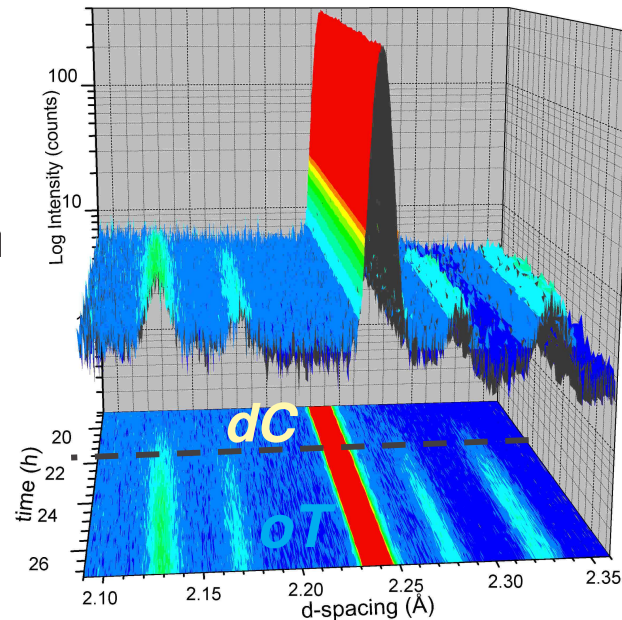
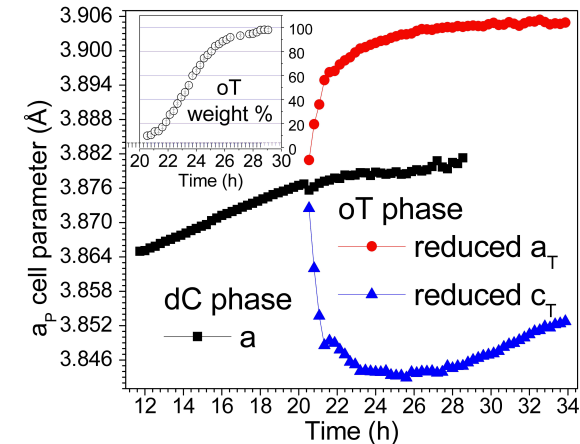
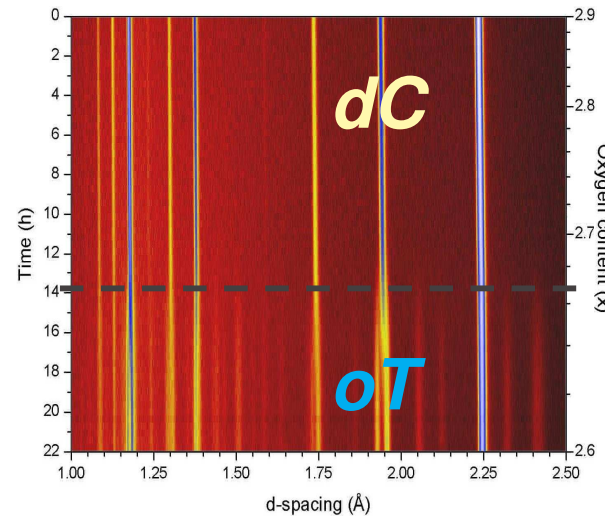
$La_{0.1}Sr_{0.9}MnO_x$: In-situ NPD experiment

- Initial interpretation:
Sample shows two vacancy ordered phases similar to oT and oO that forms consecutively upon reduction of oxygen content of initial cubic (vacancy-disordered) dC phase.
- Reality was again more complex:
The oT and oO vacancy-ordering patterns are stable in the $La_{0.1}Sr_{0.9}MnO_x$ system. However, partial vacancy filling is observed in both cases leading to the formation of oT' - $(La_{0.1}Sr_{0.9})_5Mn_5O_{13+\delta}$ and oO' - $(La_{0.1}Sr_{0.9})_2Mn_2O_{5+\delta}$ closely related to original oT and oO .

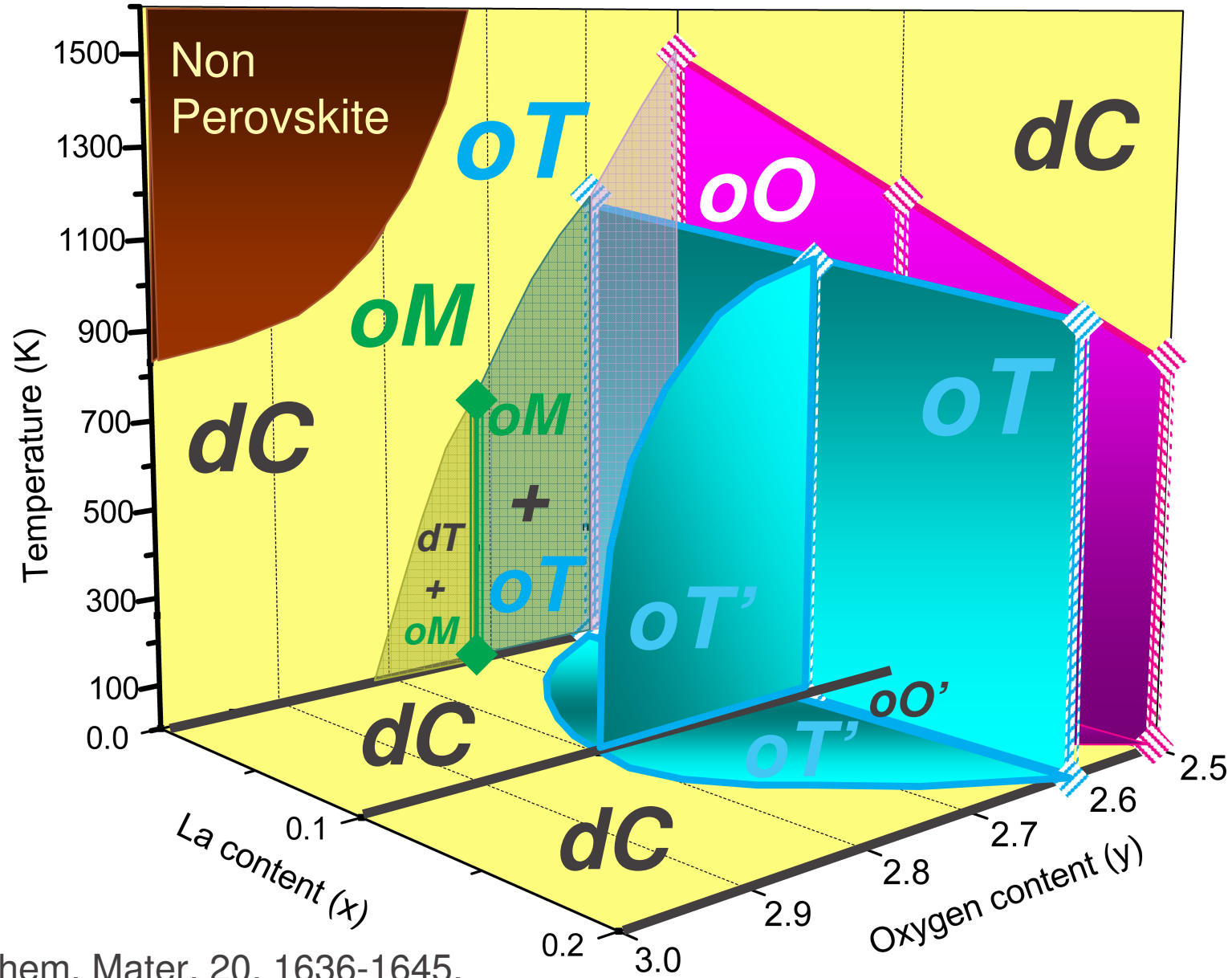
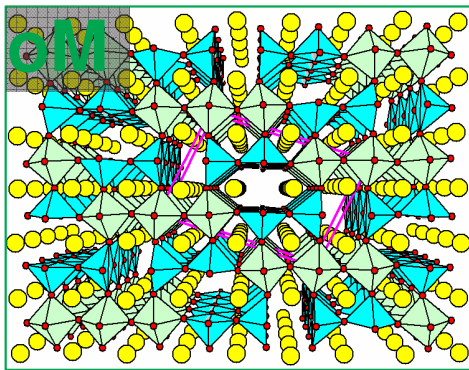
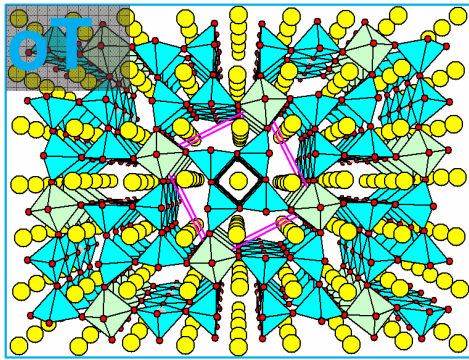
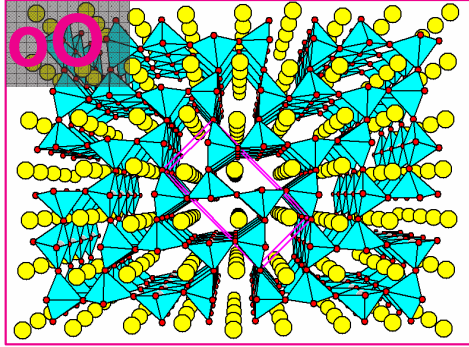


$La_{0.2}Sr_{0.8}MnO_x$: In-situ NPD experiment

- Initial (and final) interpretation: Simple cubic phase converts into ordered phase $oT-(La_{0.2}Sr_{0.8})_5Mn_5O_{13}$.
- However vacancy-ordered phase shows significant structural distortions that reduces its symmetry from tetragonal to monoclinic.
- The distortions are a consequence of a combination of orbital ordering with structural disorder.
- Further studies showed the existence of the phase $oO-(La_{0.2}Sr_{0.8})_2Mn_2O_5$



Composition-Temperature-Oxygen content phase diagram

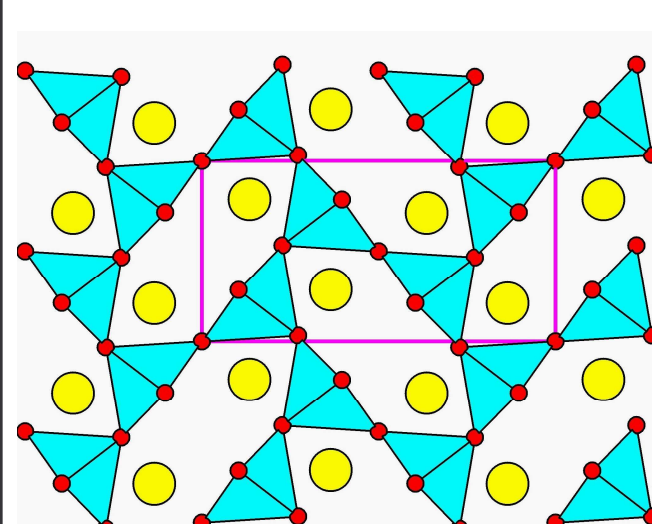


Suescun L. et al, (2008) Chem. Mater. 20, 1636-1645.

Overview

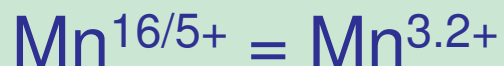
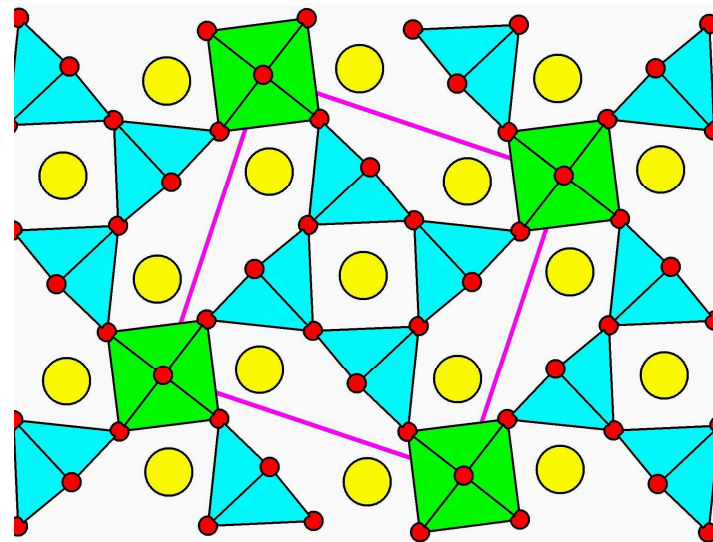
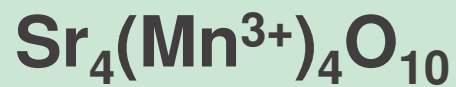
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Charge ordering in $Sr_5Mn_5O_{13}$ and $Sr_7Mn_7O_{19}$



Pyramids

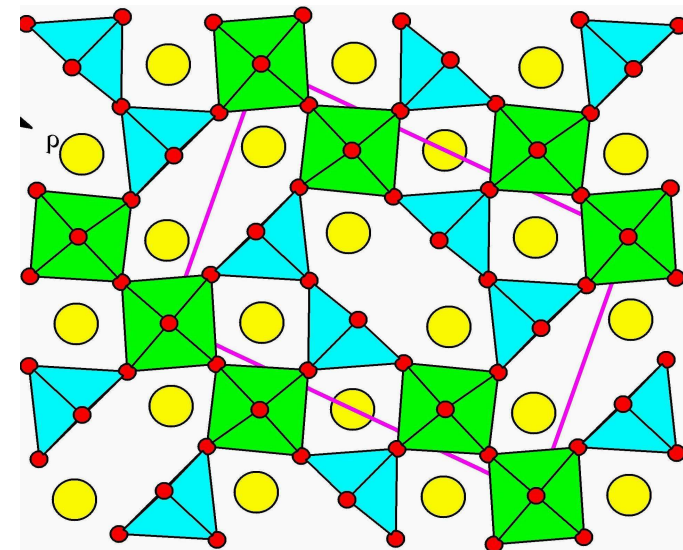
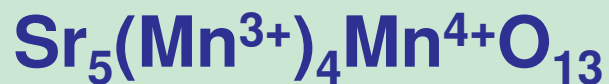
BVS: Mn1 3.08 (P)



4 Mn^{3+} P+ 1 Mn^{4+} O

BVS:

Mn1 3.81(O) Mn2 3.15 (P)

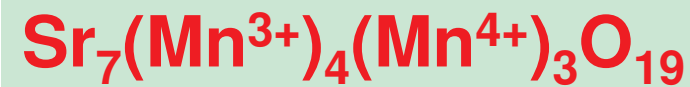


4 Mn^{3+} P+ 3 Mn^{4+} O

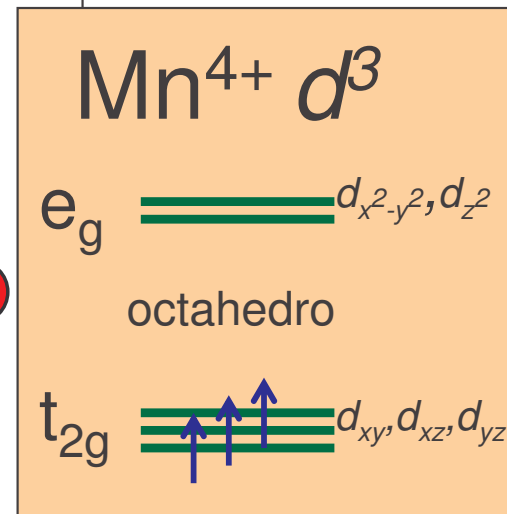
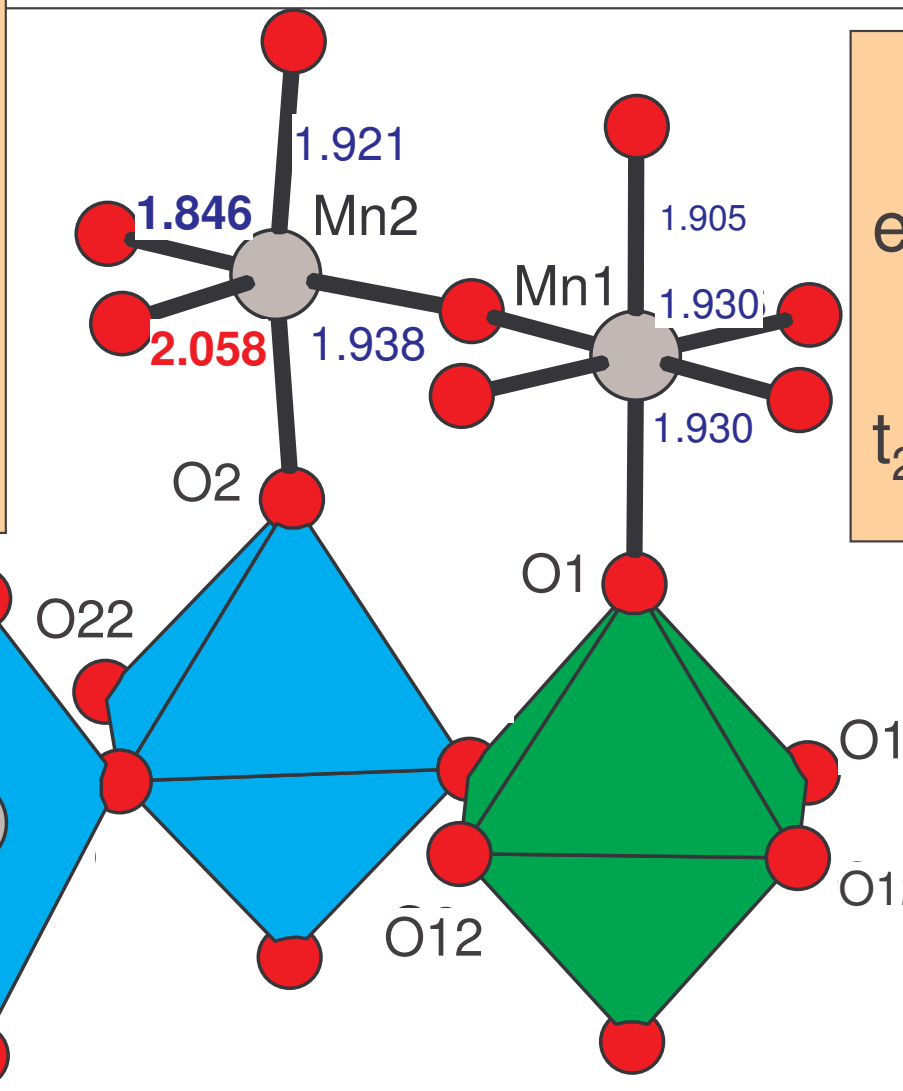
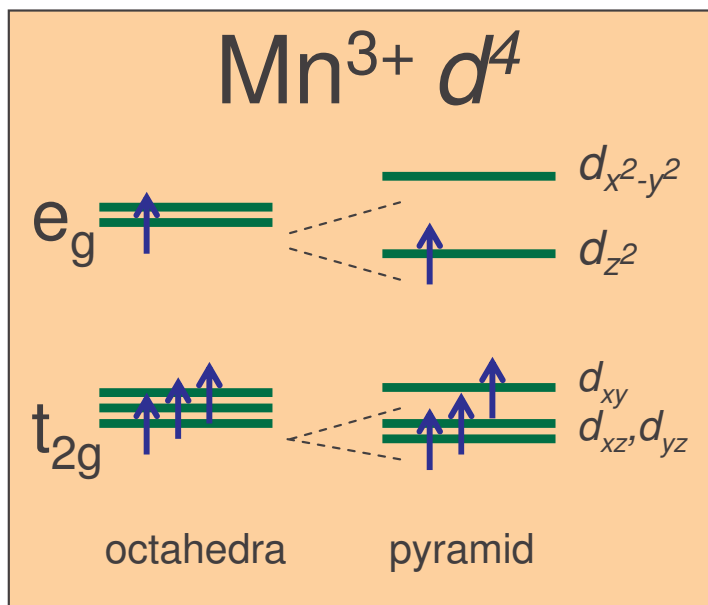
BVS:

Mn1 3.75 Mn2 4.00 (O)

Mn3 3.16 Mn4 3.18 (P)



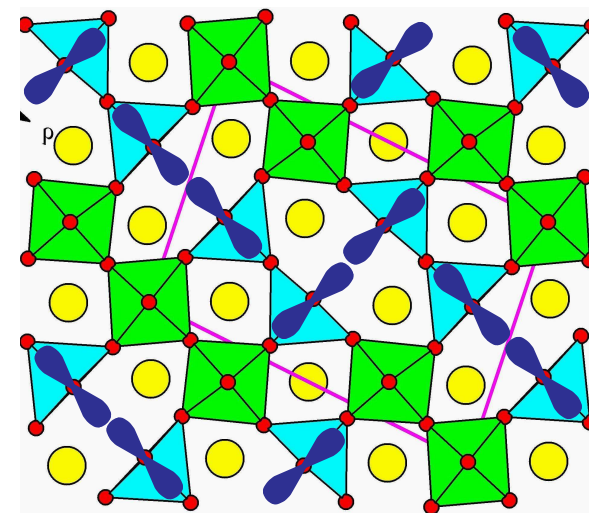
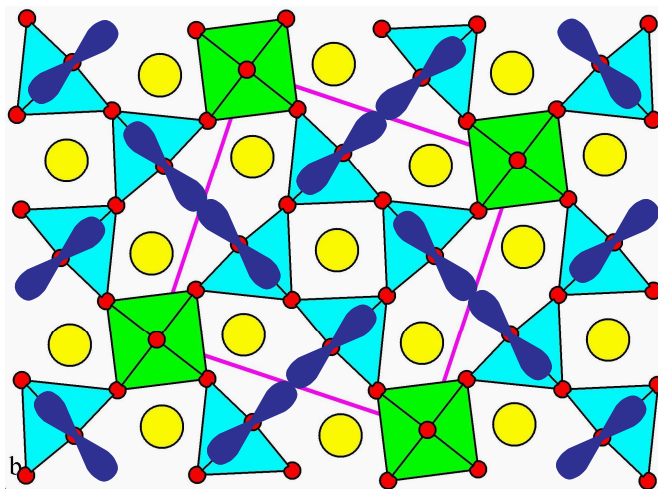
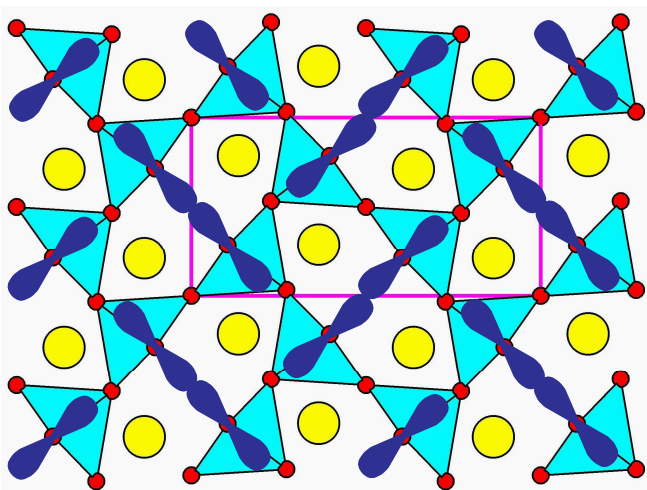
Orbital ordering in $Sr_5Mn_5O_{13}$ and $Sr_7Mn_7O_{19}$



Jahn-Teller active Mn³⁺ cations (d⁴) are located in elongated pyramids with d_{z^2} orbital along the apical direction

Non J-T Mn⁴⁺ (d³) cations are located in slightly distorted octahedra

Orbital ordering in $Sr_5Mn_5O_{13}$ and $Sr_7Mn_7O_{19}$



Mn1	
O1 (x2)	1.9198(4)
O3	1.941(3)
O2a	2.063(3)
O2e	1.875(3)

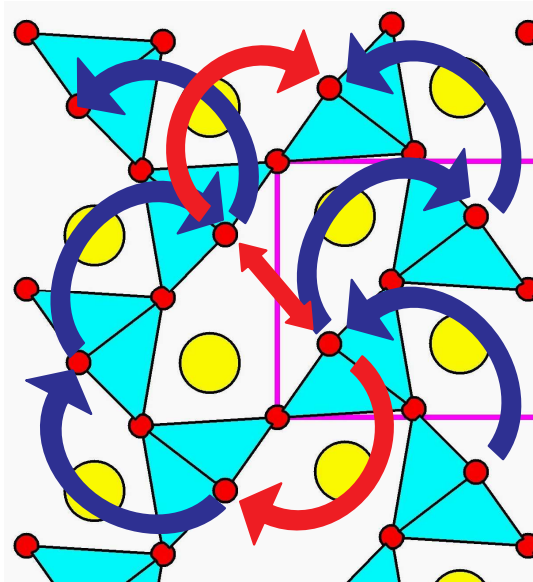
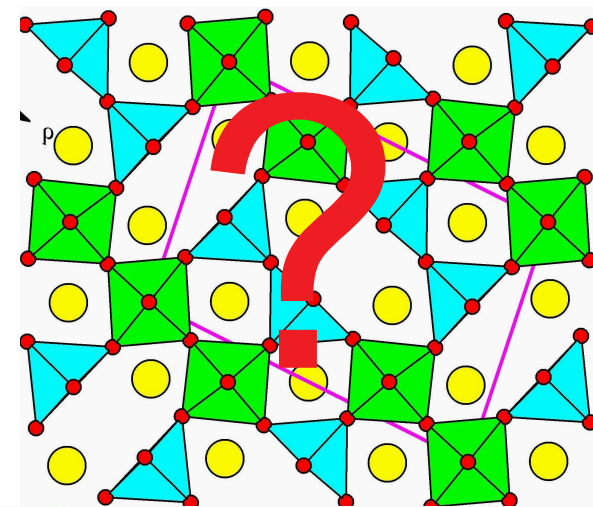
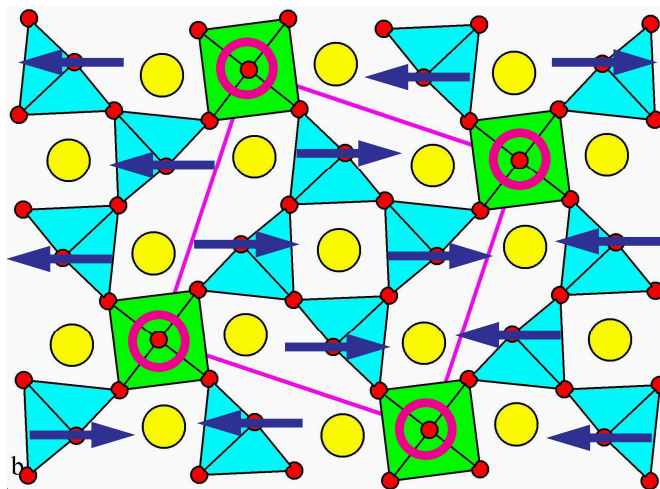
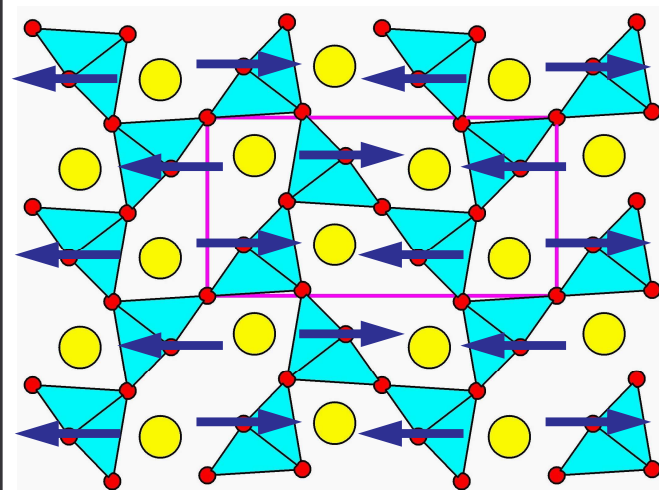


Mn2	
O2 (x2)	1.9212(5)
O12	1.938(5)
O22a	2.058(4)
O22e	1.846(4)



Mn3		Mn4	
O3 (x2)	1.914(3)	O4 (x2)	1.908(2)
O23	1.96(2)	O42	1.96(2)
O34	1.99(2)	O24	2.02(3)
O13	1.88(3)	O34	1.86(2)

Magnetic ordering in $Sr_2Mn_2O_5$ and $Sr_5Mn_5O_{13}$

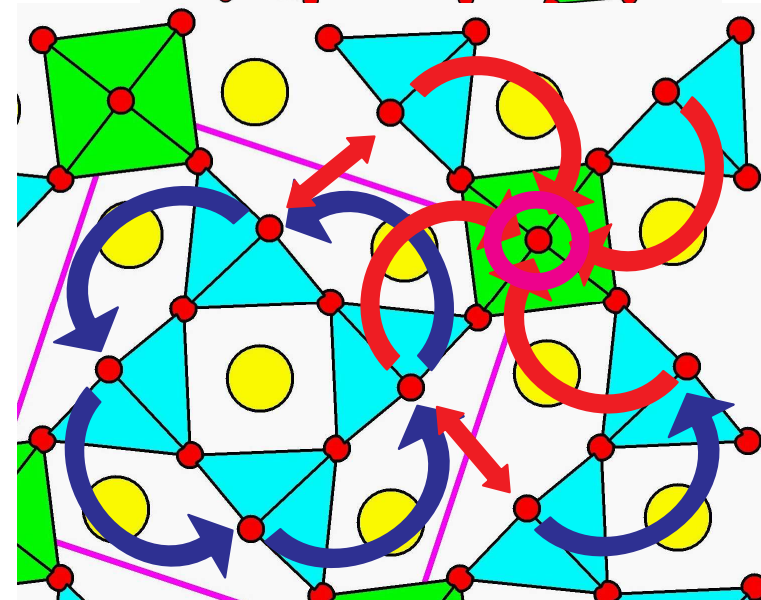


Complex exchange interactions make complex magnetic structures

Blue: Ferromagnetic

Red: Antiferromagn.

Pink: Frustration!!!!

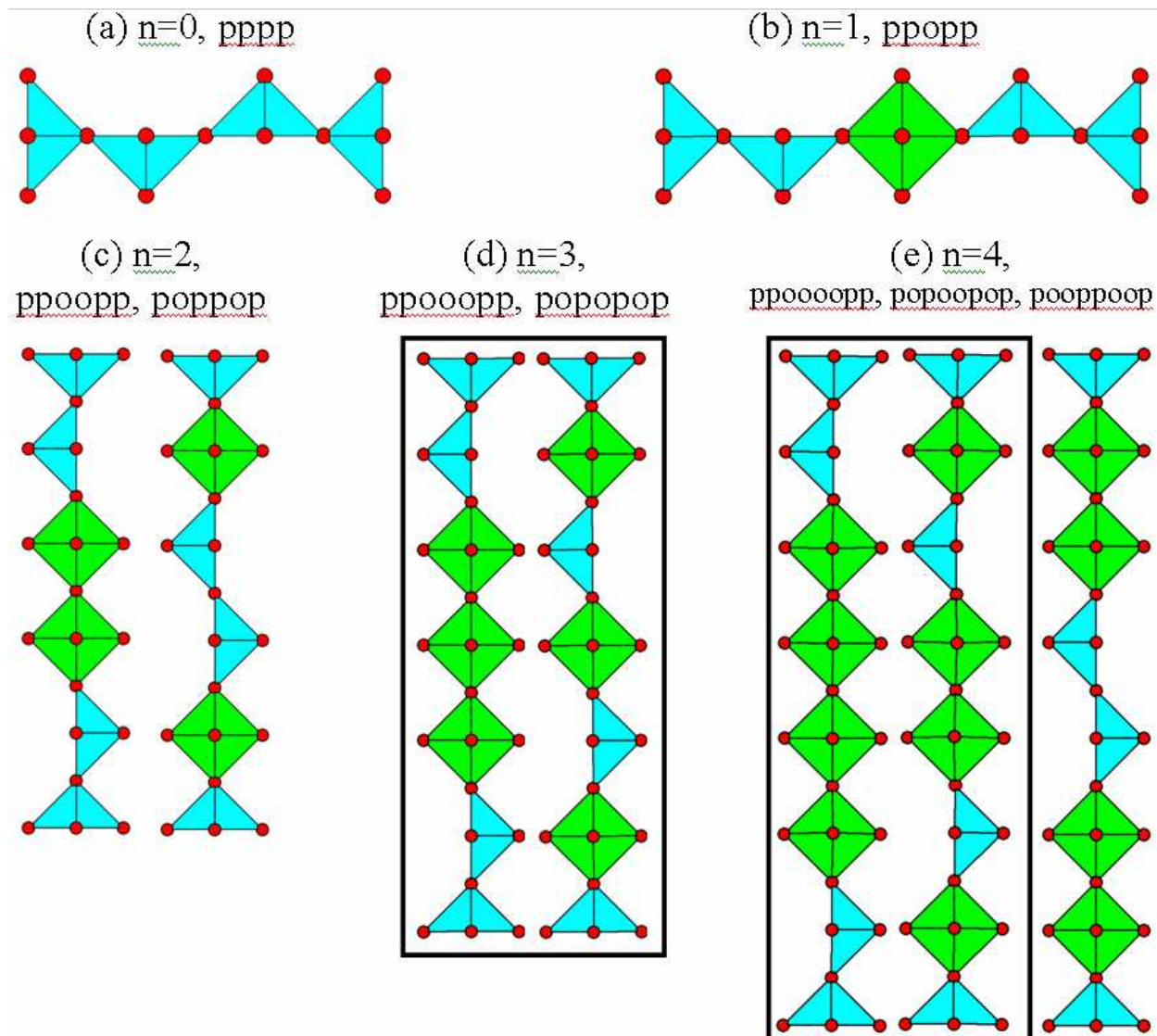


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A new homologous series $Sr_{4+n}Mn^{3+}_4Mn^{4+}_nO_{10+3n}$

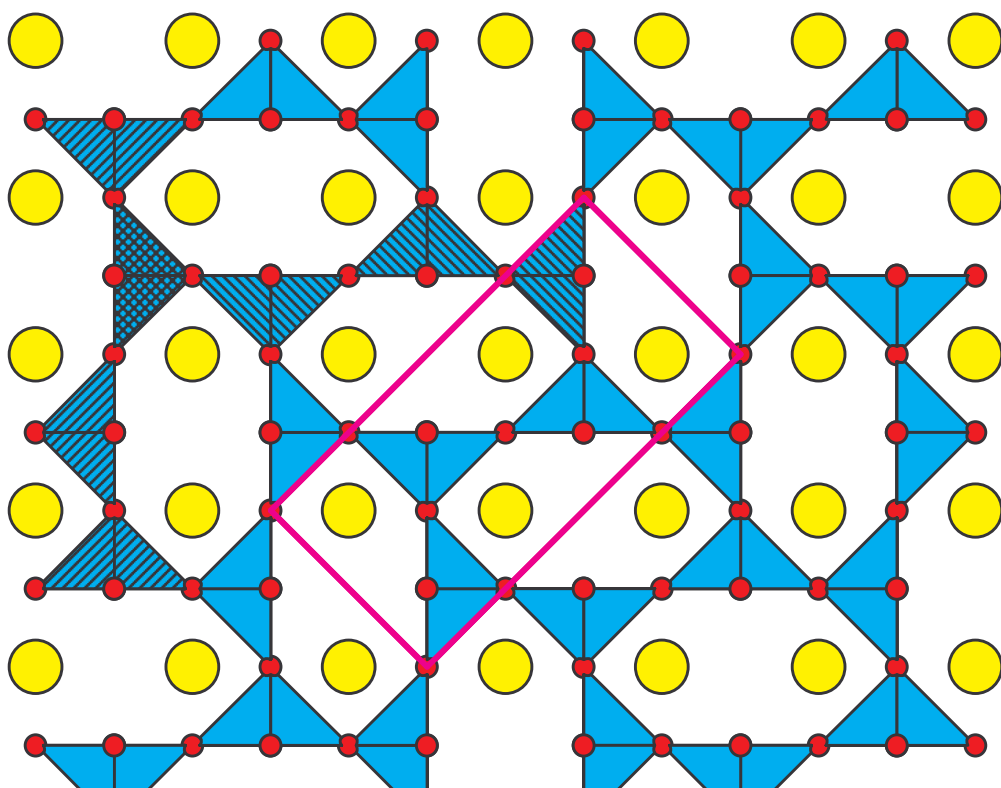
- Building blocks formed by 4 pyramids and n octahedra can be used to generate compounds in the series $Sr_{4+n}Mn^{3+}_4Mn^{4+}_nO_{10+3n}$
- Members of the series observed to date (n=0, 1 and 3) are formed by symmetrical blocks (the building block displays $2/m$ symmetry)
- For certain values of n different building blocks lead to different structures, but in some cases a unique pattern is formed that combines two blocks (boxes).
- No compound corresponding to non-unique structural models have been observed so far in the $SrMnO_x$ system.



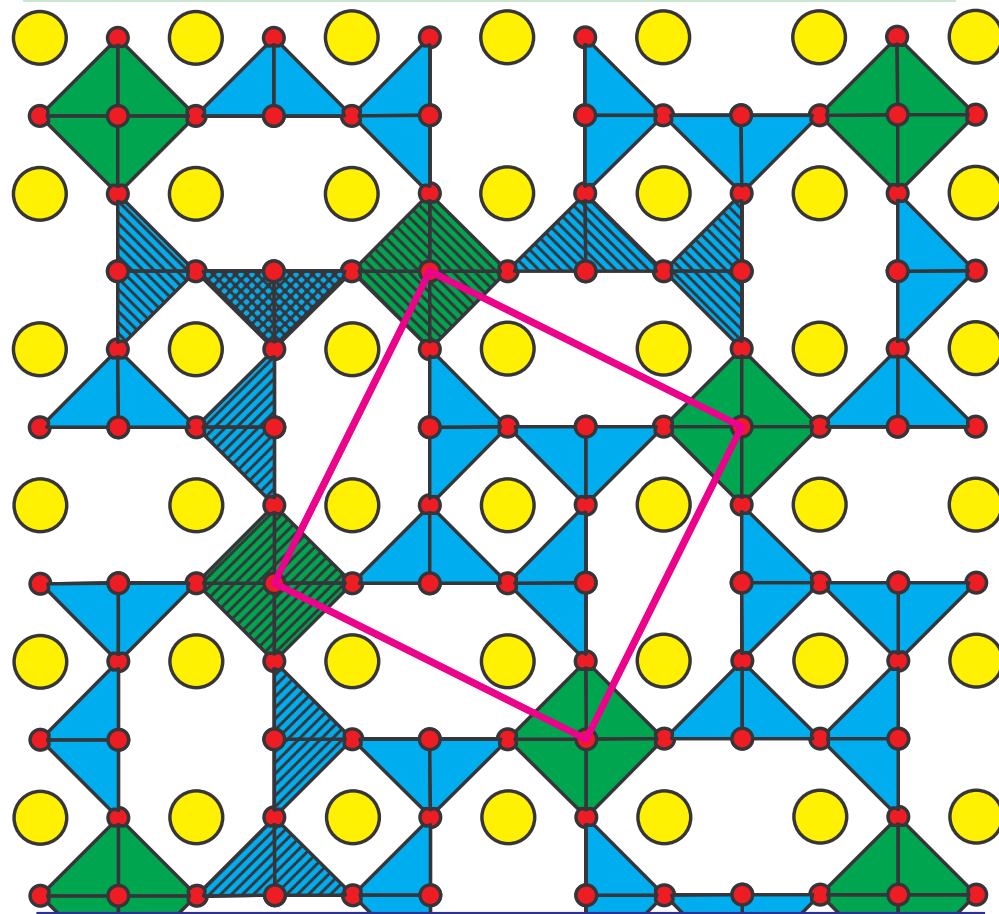
Suescun L. & Dabrowski B. (2008) Acta Crystallographica Section B, 64, 177-186.

A new homologous series $Sr_{4+n}Mn^{3+}_4Mn^{4+}_nO_{10+3n}$

$Sr_4Mn_4O_{10}$ structure can be built based on $n=0$ block formed by 4 pyramids oriented along $+x,-y,+y,-x$ along the lattice constants (or l.c. of them). The 4-pyramid pattern forms automatically in the orthogonal direction.

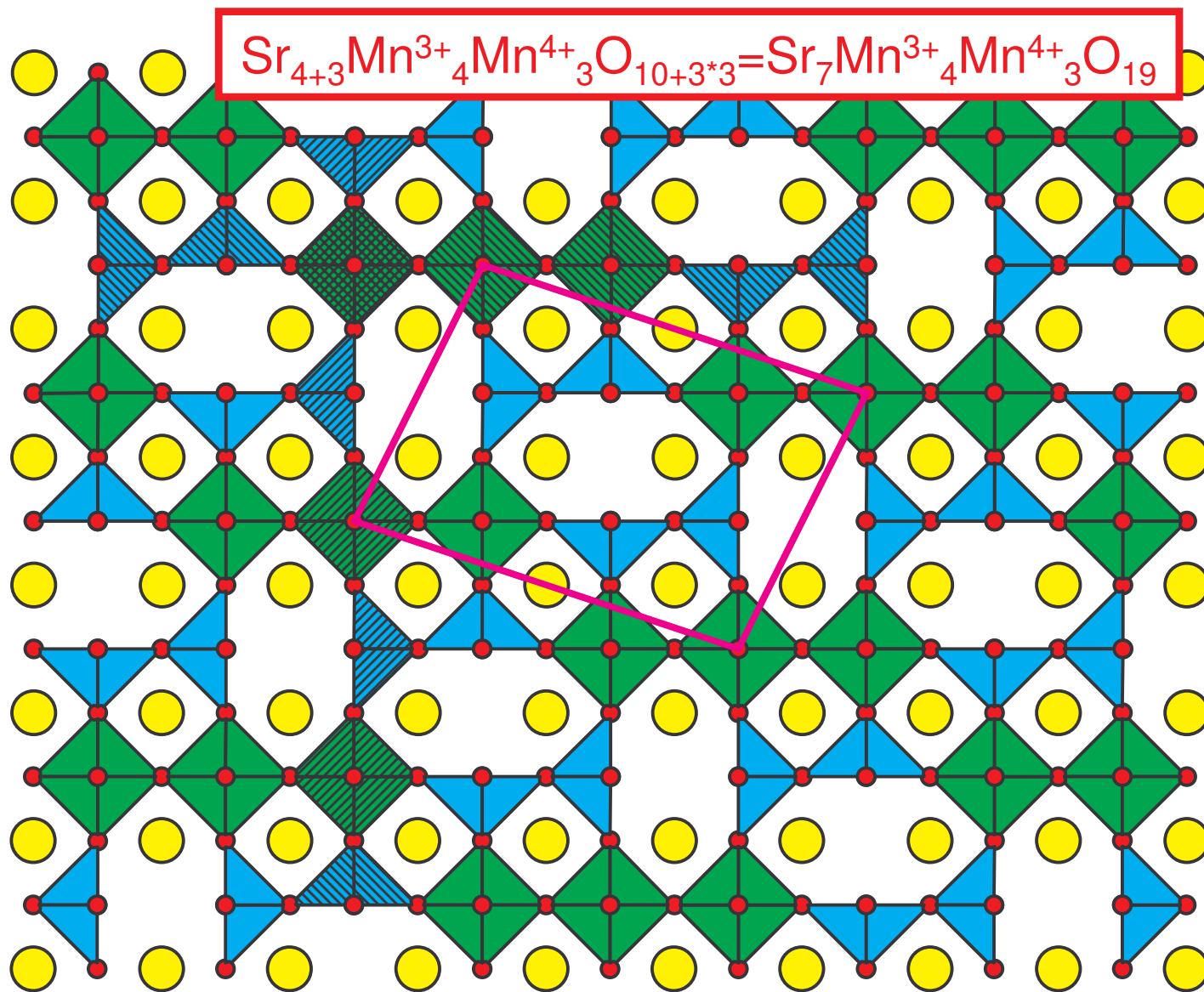


$Sr_5Mn_5O_{13}$ structure can be built based on the $n=1$ block containing 4 pyramids with one octahedron in the center



A new homologous series $Sr_{4+n}Mn^{3+}_4Mn^{4+}_nO_{10+3n}$

$Sr_7Mn_7O_{19}$ structure can also be built in an analogous manner using a block containing 4 pyramids and 3 octahedra. If the three octahedra are located in the center of the array and the horizontal direction is chosen to build the structure another possible block with alternating pyramids and octahedra is formed in the orthogonal (vertical) direction and viceversa. Both blocks display a symmetrical pyramid/octahedra arrangement.



A new homologous series $Sr_{4+n}Mn^{3+}_4Mn^{4+}_nO_{10+3n}$

- Predicted compounds in the $Sr_{4+n}Mn^{3+}_4Mn^{4+}_nO_{10+3n}$ homologous series

n	x in ABO_x	Pyramid/octahedra	Cell parameters† (a, b, c, γ, V)	Space group (ideal)†
0	2.5 ✓	pppp	$2^{1/2}a_P \ 2(2)^{1/2}a_P \ a_P \ 4V_P$	<i>Pbam</i> (55)
1	2.6 ✓	ppopp	$5^{1/2}a_P \ 5^{1/2}a_P \ a_P \ 5V_P$	<i>P4/m</i> (83)
2	2.667	ppoopp poppop	$2a_P \ 10^{1/2}a_P \ a_P \ \gamma = 108.4 \ 6V_P$ $2^{1/2}a_P \ 3(2)^{1/2}a_P \ a_P \ 6V_P$	<i>P2/m</i> (10) <i>Pbmm</i> (<i>Pmma</i> , 51)
3	2.714‡ ✓	ppooopp/popopop	$5^{1/2}a_P \ 10^{1/2}a_P \ a_P \ \gamma = 98.2 \ 7V_P$	<i>P2/m</i> (10)
4	2.75	ppooooopp/popooooop pooppoop	$2(2)^{1/2}a_P \ 10^{1/2}a_P \ a_P \ \gamma = 116.6 \ 8V_P$ $2^{1/2}a_P \ 4(2)^{1/2}a_P \ a_P \ 8V_P$	<i>P2/m</i> (10) <i>Pbam</i> (55)

- Samples with composition $SrMnO_{2.667}$ (corresponding to $n=2$ member of the series) have been obtained as a mixture of $Sr_5Mn_5O_{13}$ ($SrMnO_{2.6}$, $n=1$) and $Sr_7Mn_7O_{19}$ ($SrMnO_{2.714}$, $n=3$).
- Samples with compositions $SrMnO_x$ $2.7 < x < 2.8$ (oxygen content corresponding to $n=3$, $n=4$ and $n=5$) are a mixture of $Sr_7Mn_7O_{19}$ ($SrMnO_{2.714}$, $n=3$) and a vacancy-disordered phase with approximate composition $SrMnO_{2.82}$ (unpublished).

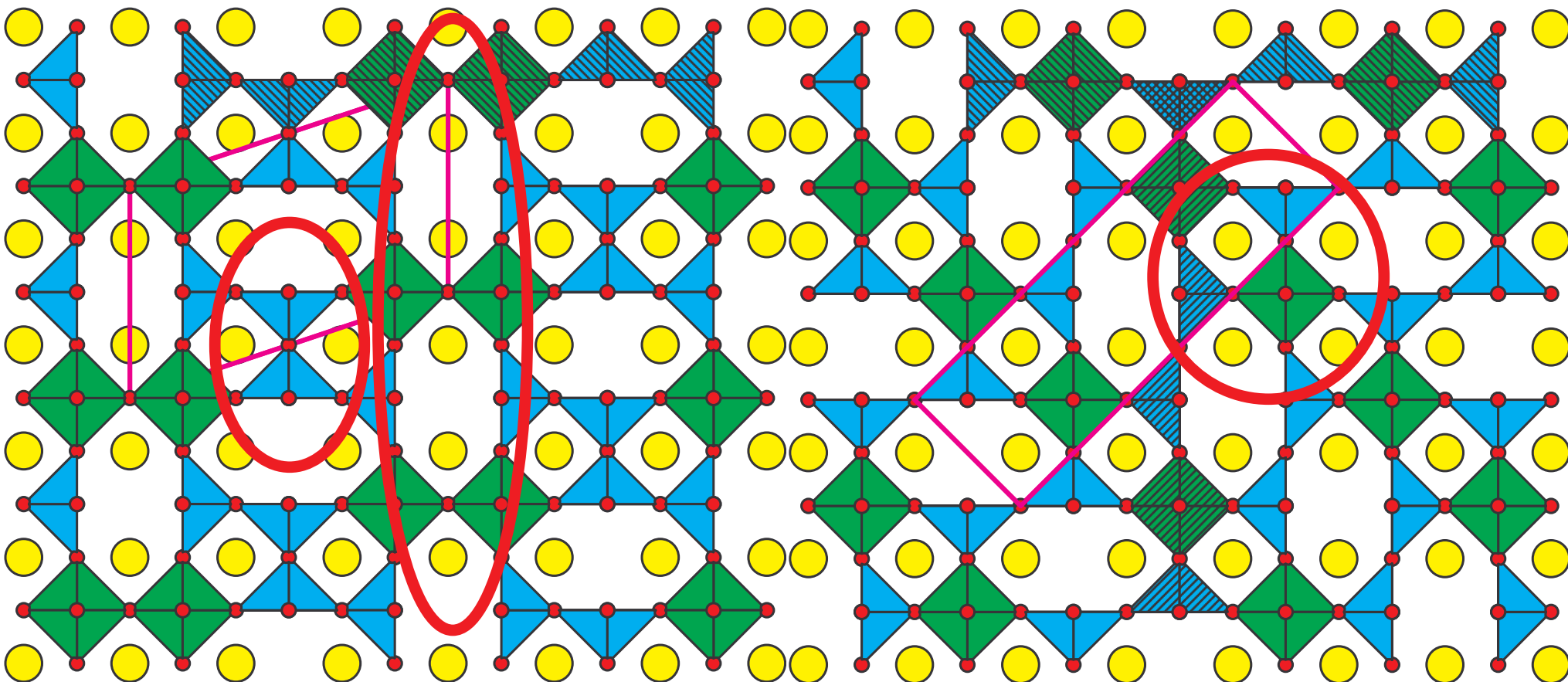
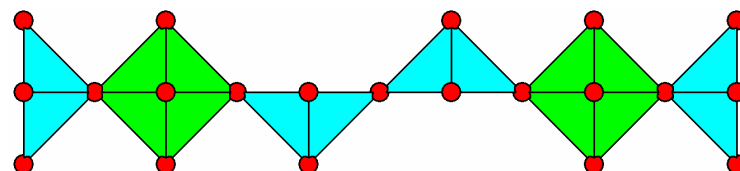
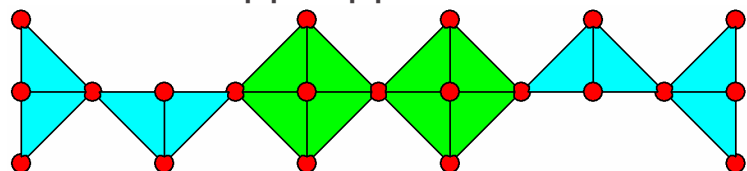
Suescun L. & Dabrowski B. (2008) *Acta Crystallographica Section B*, 64, 177-186.

A new homologous series $Sr_{4+n}Mn^{3+}_4Mn^{4+}_nO_{10+3n}$

- Predicted compounds with $n=2$ ($Sr_6Mn^{3+}_4Mn^{4+}_2O_{16}$) show unfavorable features:

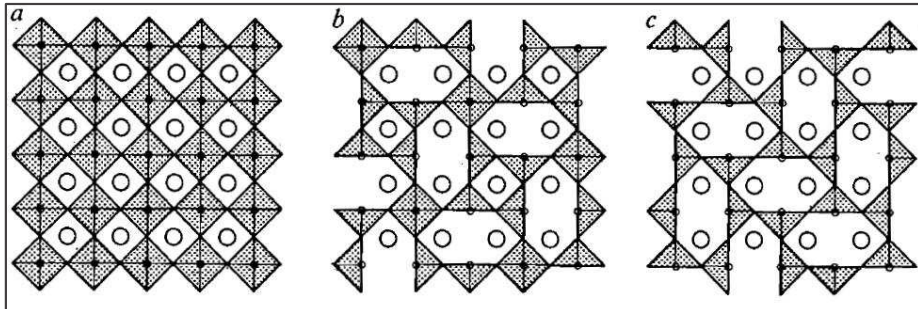
ppoop-block

poppop-block



Other systems showing $A_{4+n}B_{4+n}O_{10+3n}$ -type ordering:

$LaCuO_x$ & $NdCuO_x$ systems



- **N=0** observed for d^9 Cu^{2+} (pyramids)
- **N=1** observed for d^9 Cu^{2+} and d^8 Cu^{3+} (octahedra). Large monoclinic distortion observed
- No phases with **N>2** observed

Bringley et al, *Letters to Nature* (1990) 347, 263-265

Chen et al, *Inorg Chem.* (1995) 34, 2077-2083.

$CaMnO_x$ system

- $Ca_2Mn_2O_5$ (**N=0**) structure has been determined
- HREM and ED studies of $CaMnO_{2.667}$ and $CaMnO_{2.75}$ have shown formation of local structures with unit cells compatible with those proposed for **N=2** ($Ca_6Mn_6O_{16}$) and **N=4** ($Ca_8Mn_8O_{22}$) members of the series respectively
- A neutron powder diffraction of $CaMnO_{2.75}$ was inconclusive possibly due to the coexistence of multiple ordering arrangements

Poepelmeier et al *J. Solid State Chem.* (1982) 45, 79-79.

Reller et al *Proc. R. Soc. Lond. A* (1984) 349, 223-241.

Chiang & Poepelmeier, *Mater Lett.* (1991) 12, 102-108.

Size and charge of A-site cation appears to play a key role in the stabilization of different structural patterns.

Perspectives

- Settle down in Uruguay
- Get all that unprocessed data published
- Visit Argonne before April 2009
- Continue performing experiments
 - In-situ NPD experiments at SNS
 - In-situ Synchrotron X-ray diffraction experiments at Brazilian Synchrotron Laboratory

Contributors to this work

- Bogdan Dabrowski & J.D. Jorgensen/Ray Osborn (supervisors)
- At Materials Science Lab. – Physics Department - NIU:
 - Konrad Świerczek, Omar Chmaissem (also at ANL)
 - Jim Mais, Steve Remsen, Ben Stillwell (synthesis, transport)
 - Stan Kolesnik (magnetism)
- At IPNS
 - SEPD staff: Simine Short, Joe Fieramosca, Ryoji Kiyonagi, Bob von Dreele
 - GPPD staff: Evan R. Maxey, Jim Richardson (RIP)
- At APS
 - Yang Ren (11 ID-C), Brian H. Toby, Jun Wang
- At Universidad de la República, Facultad de Química (Montevideo, Uruguay)
 - Alvaro W. Mombrú, Ricardo Faccio, Helena Pardo, Luciana Fernández

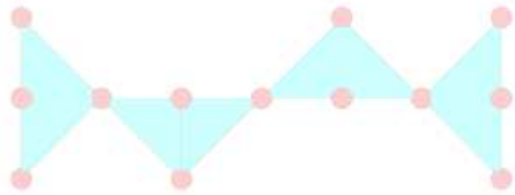
Use of the Advanced Photon Source and Intense Pulsed Neutron Source was supported by the U. S. Department of Energy, Office of Science, Office of Basic Energy Sciences, under Contract No. DE-AC02-06CH11357. Work at NIU was supported by the NSF Grant No. DMR-0302617, the U.S. Department of Education and the U.S. Department of Transportation.

In memoriam

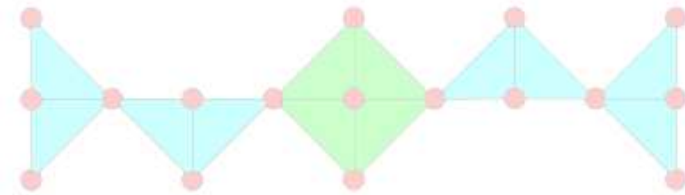


James D. Jorgensen, 1948 - 2006

(a) $n=0$, pppp



(b) $n=1$, ppopp



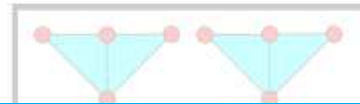
(c) $n=2$,

ppooppp, popppop



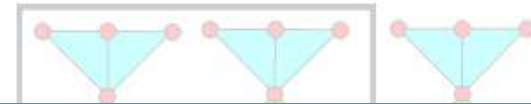
(d) $n=3$,

ppooooopp, popopop



(e) $n=4$,

ppooooopp, popoopop, pooppoop



Questions? Comments?

